

New Tools For Texture Research

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Montreal, Canada

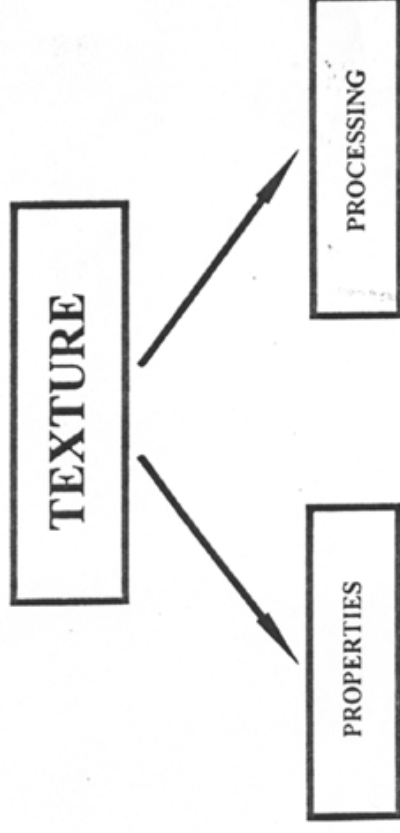
For demonstration of texture analysis software, go to

www.resmat.com

- F. Czerwinski, D. Li, K. Lee, P. Blandford,
- Hualong Li, Zhang Li, D. Hinz, R. Rajmohan. V. Gertzman, A. Zhilyaev

Acknowledgments

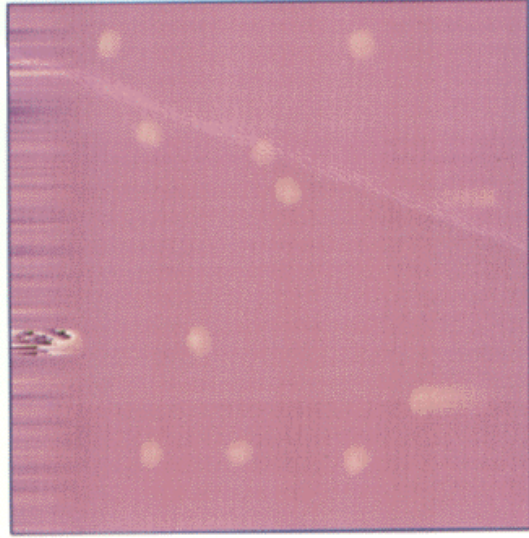
Why Study Texture in Films?



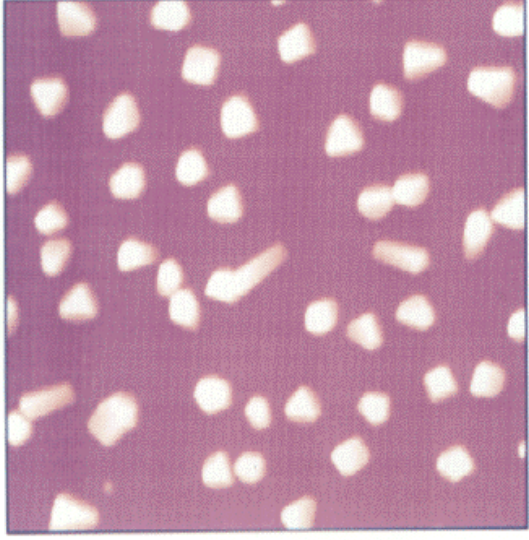
PROPERTIES AND APPLICATIONS PROCESSING

- semiconductor films
- insulating and conductive films
- films for magnetic recording
- magnetic recording heads
- optical recording
- superconductive films
- magnetic multilayers
- diffusional, protective and thermal coatings
- tribology of films and coatings
- diamond films
- physical vapour deposition
- chemical vapour deposition
- sol-gel
- electroplating
- annealing
- laser modification
- ion implantation

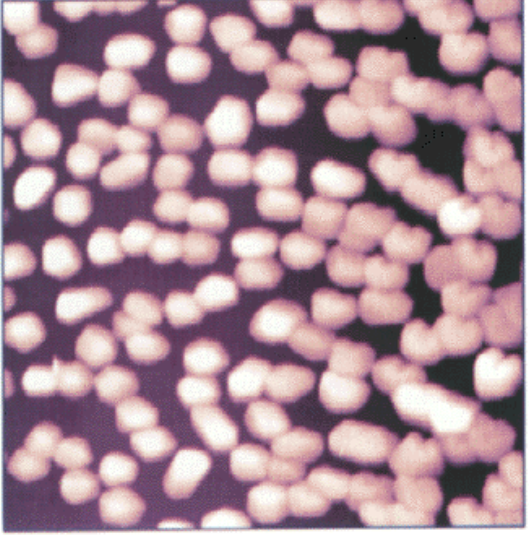
tmquen48.021



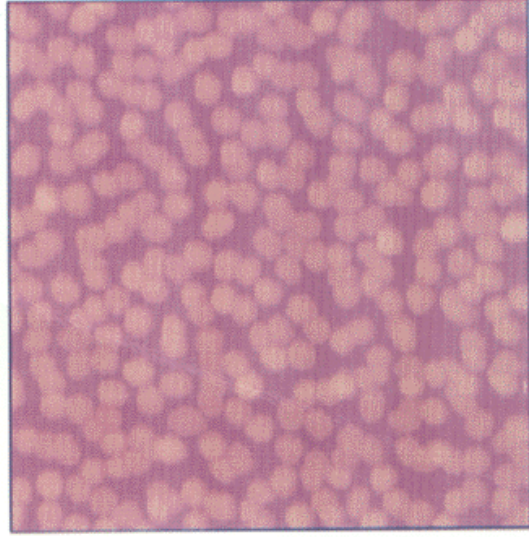
tmquen44.001



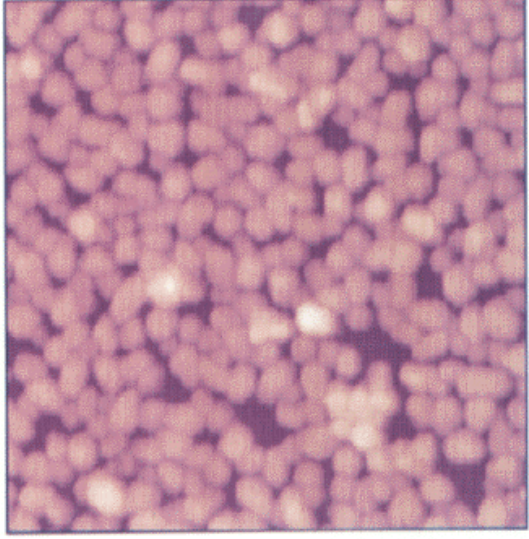
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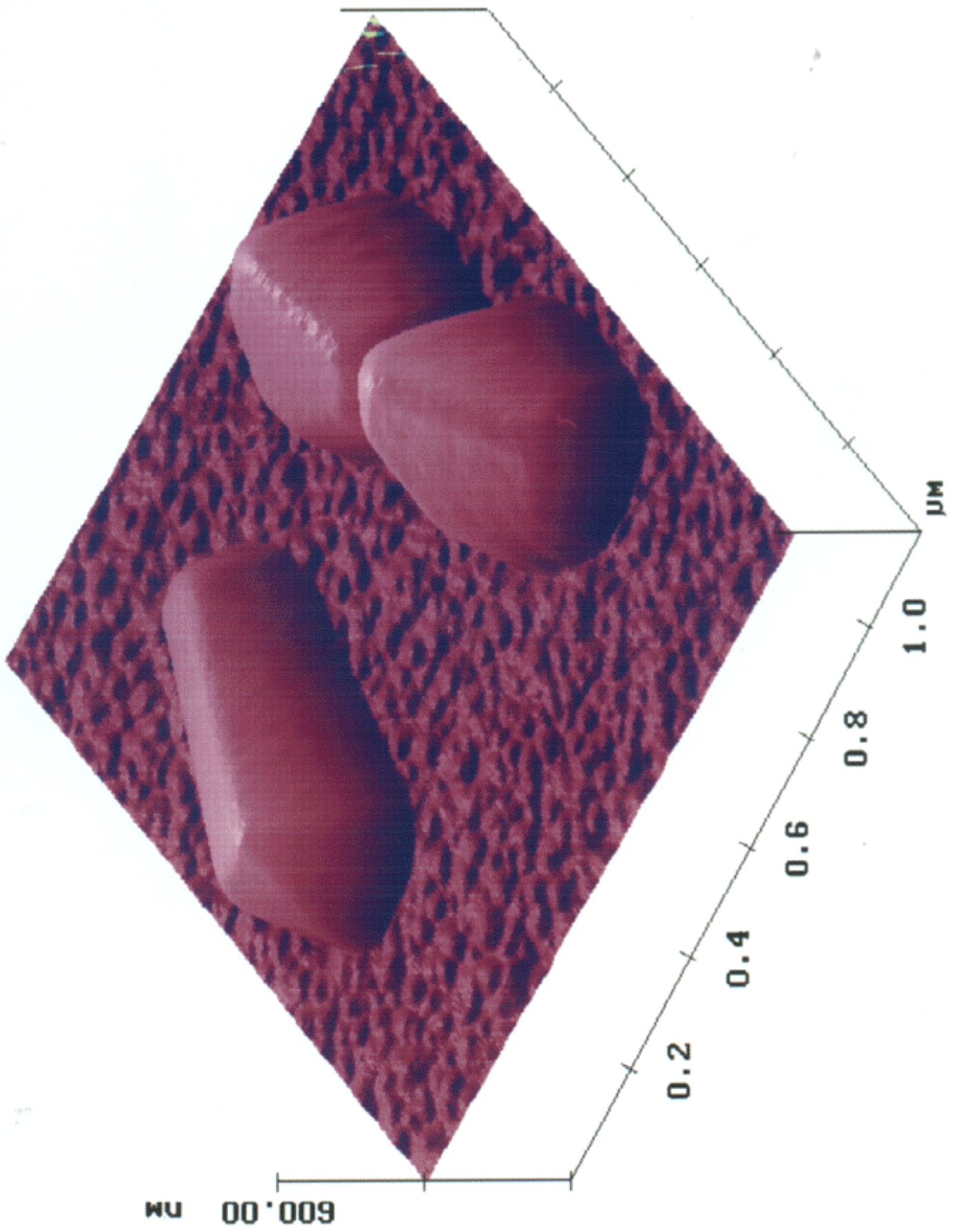


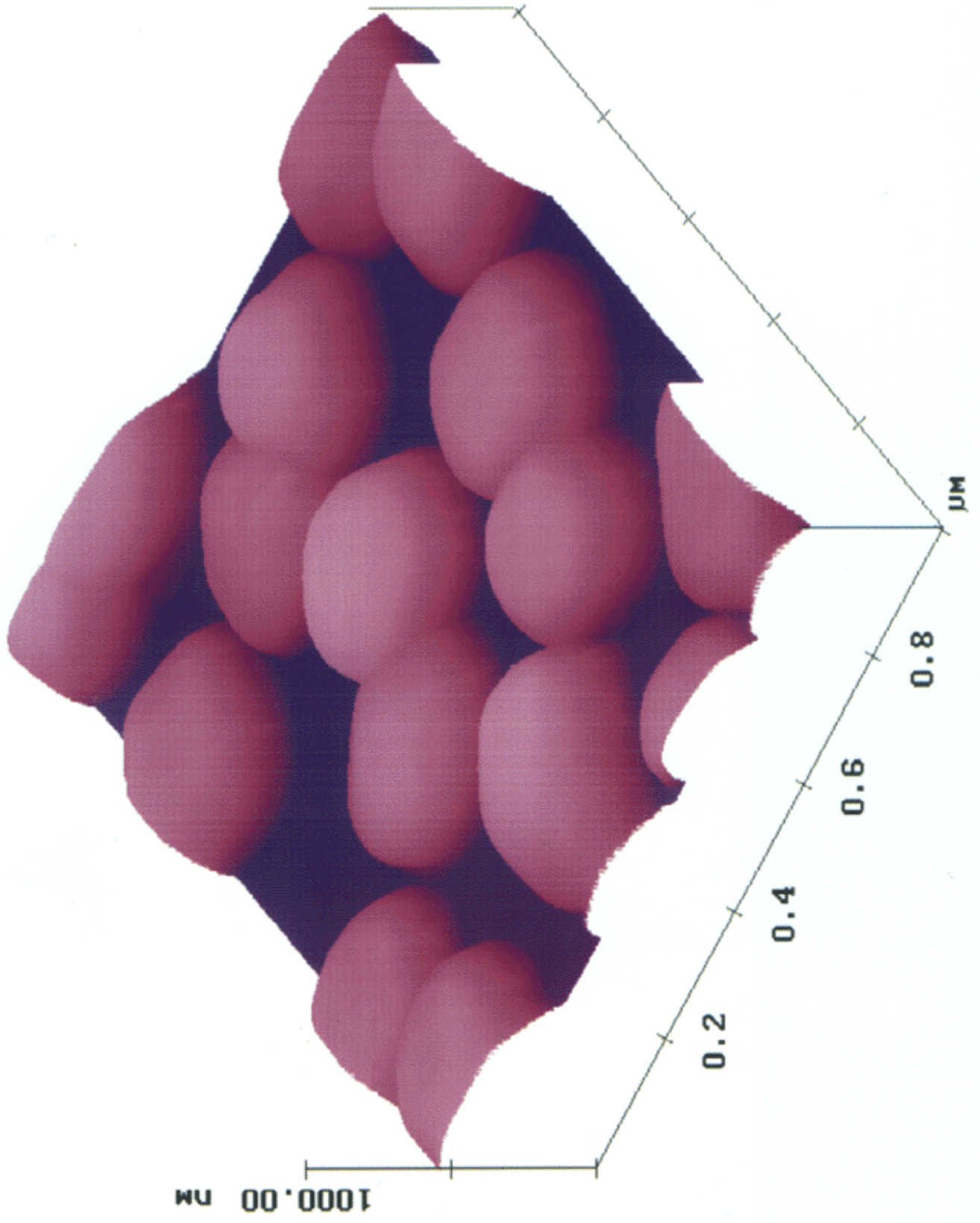
tmquen46.001



tmquen51.002

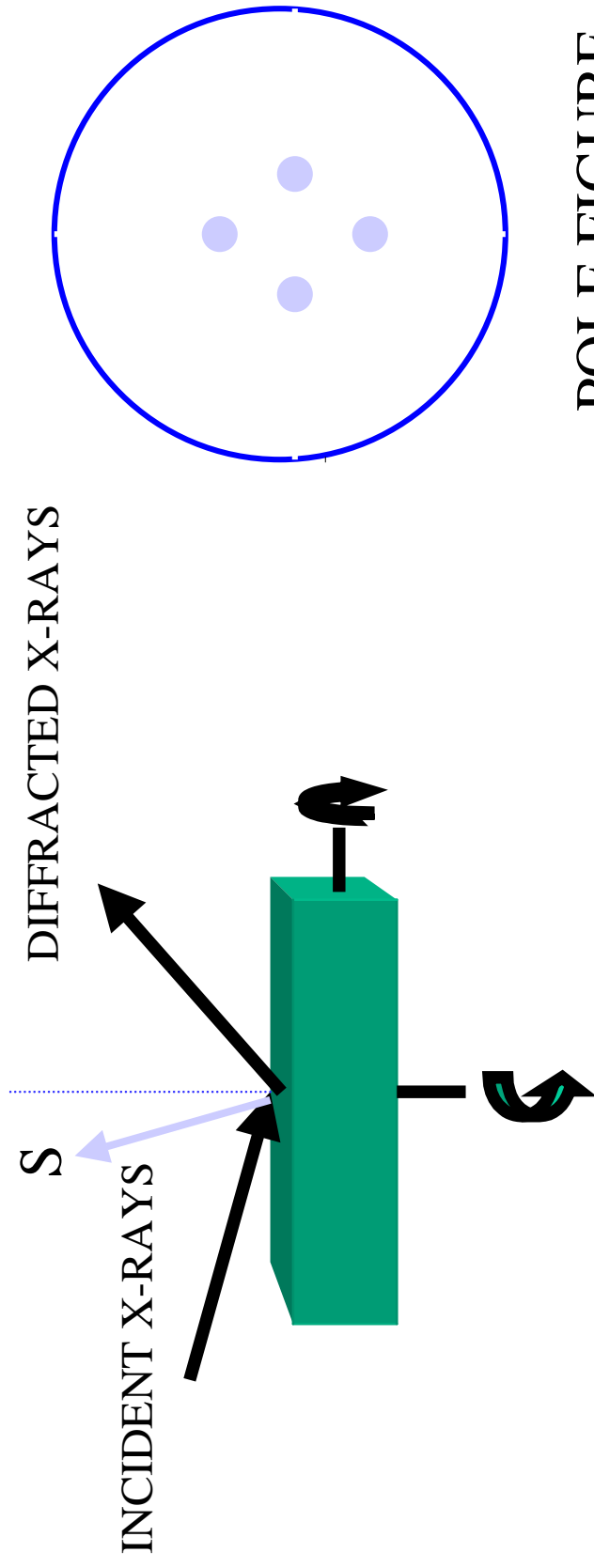




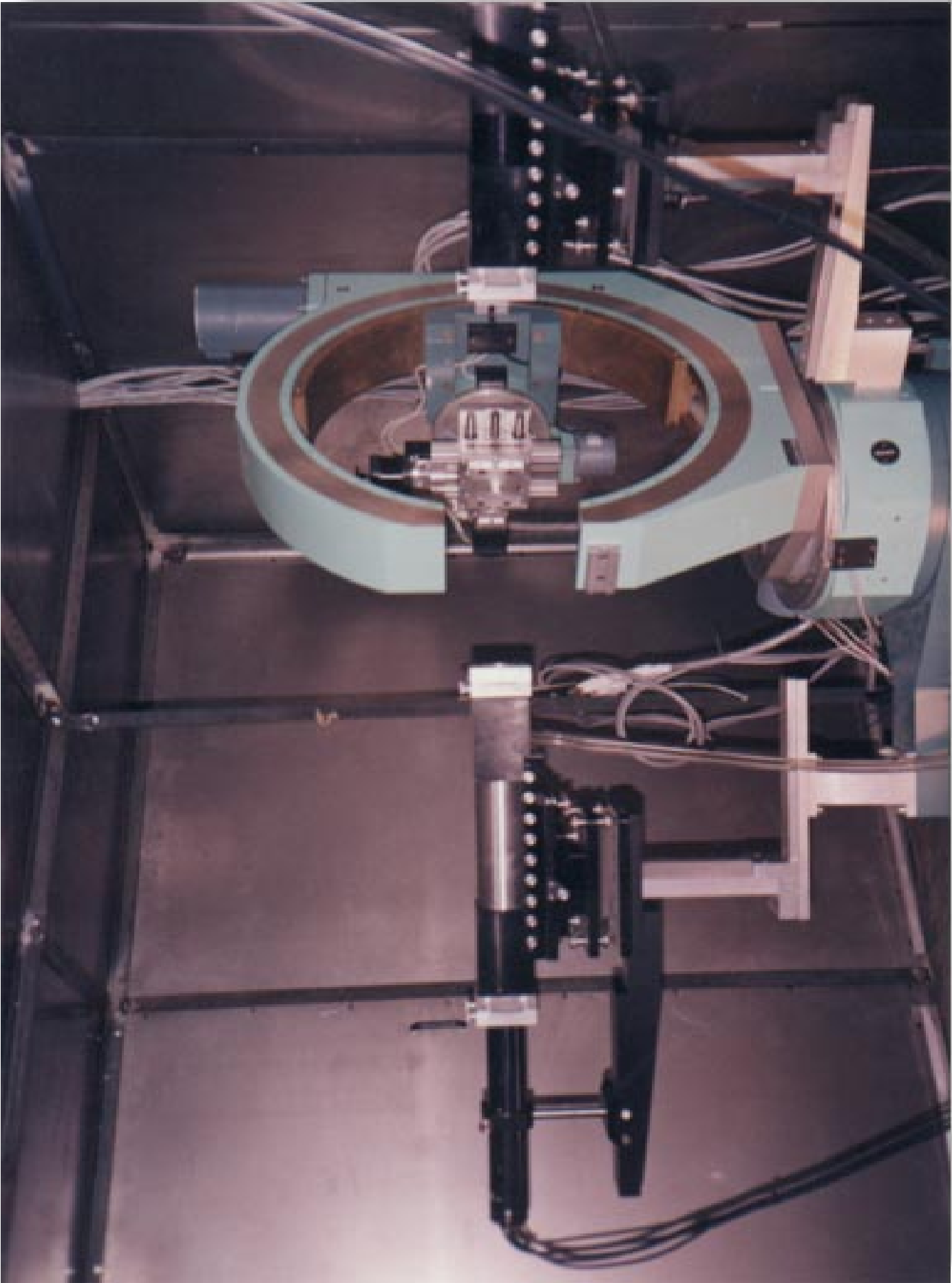


TEXTURE MEASUREMENTS

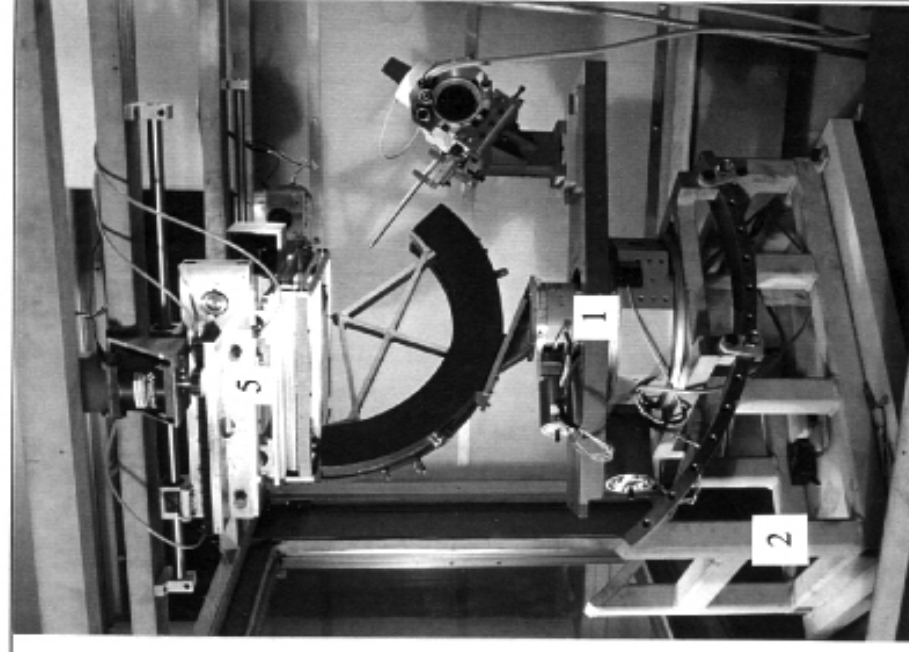
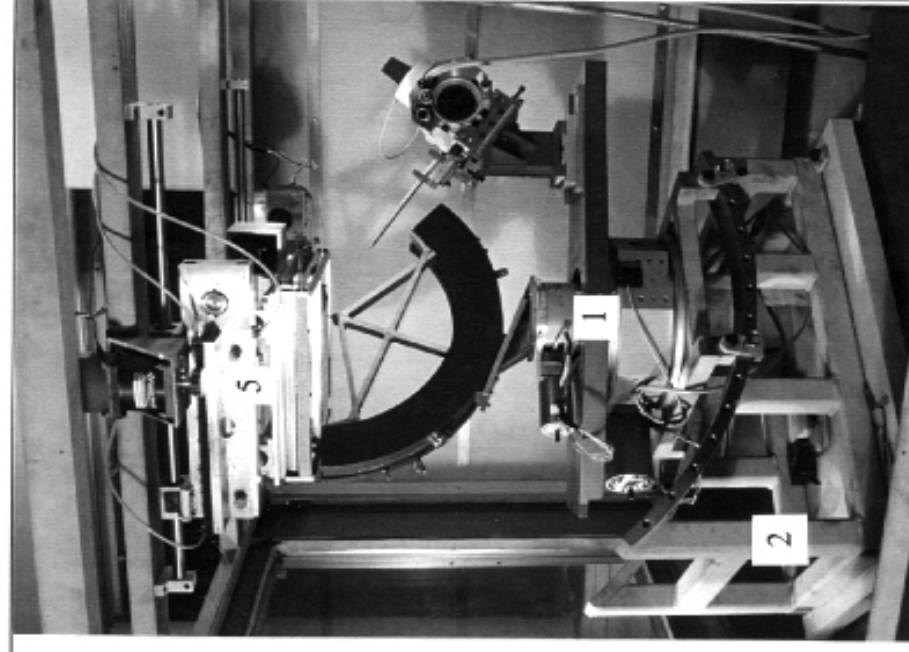
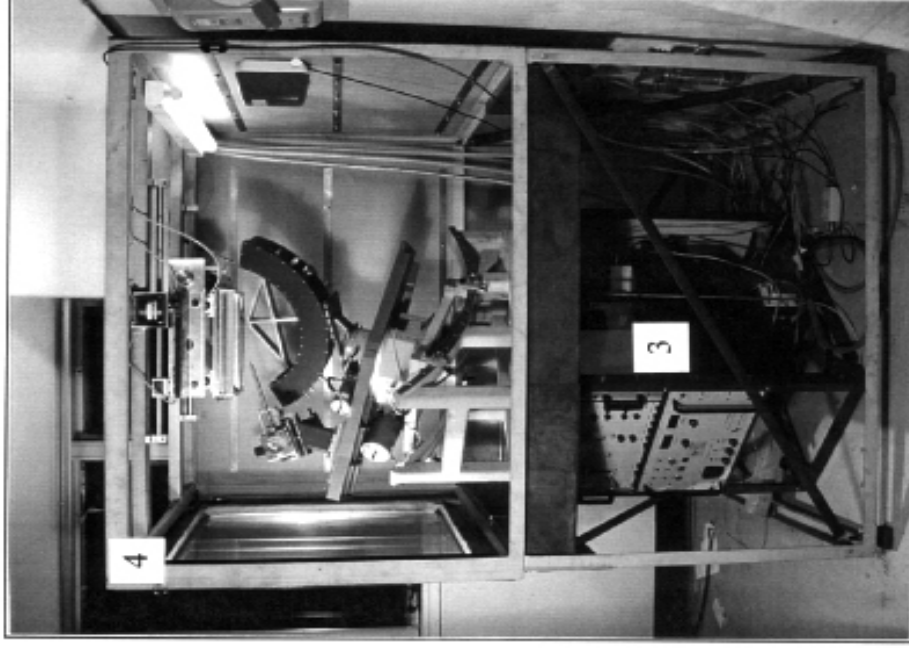
- GRAZING INCIDENCE GEOMETRY



POLE FIGURE

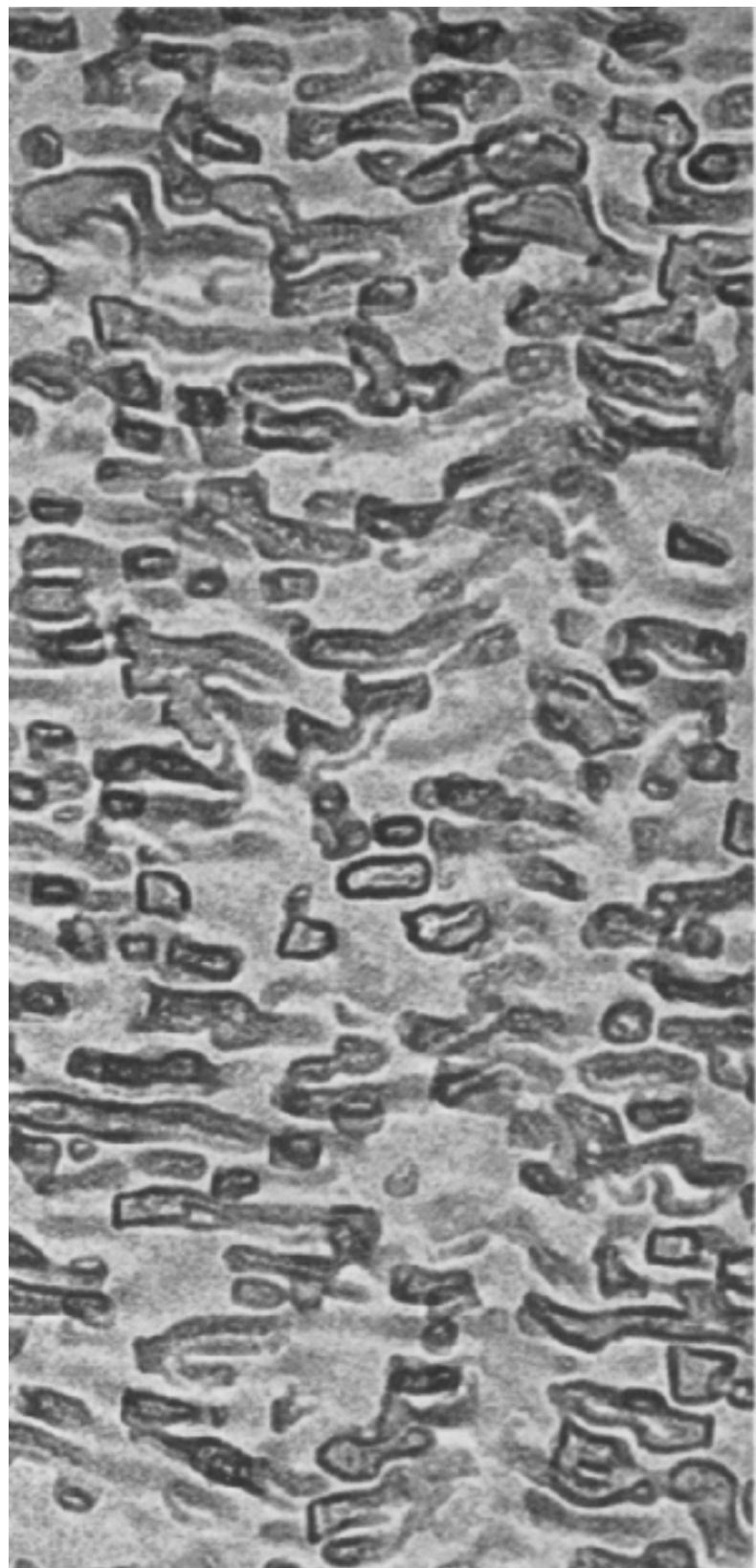


Photographs of the laboratory facility built to test
methods for on-line texture analysis



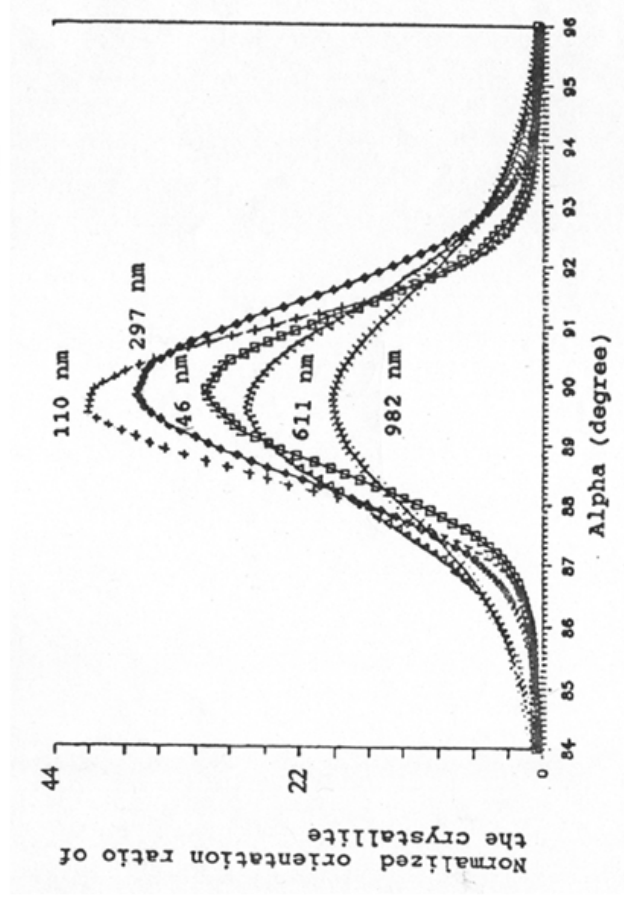
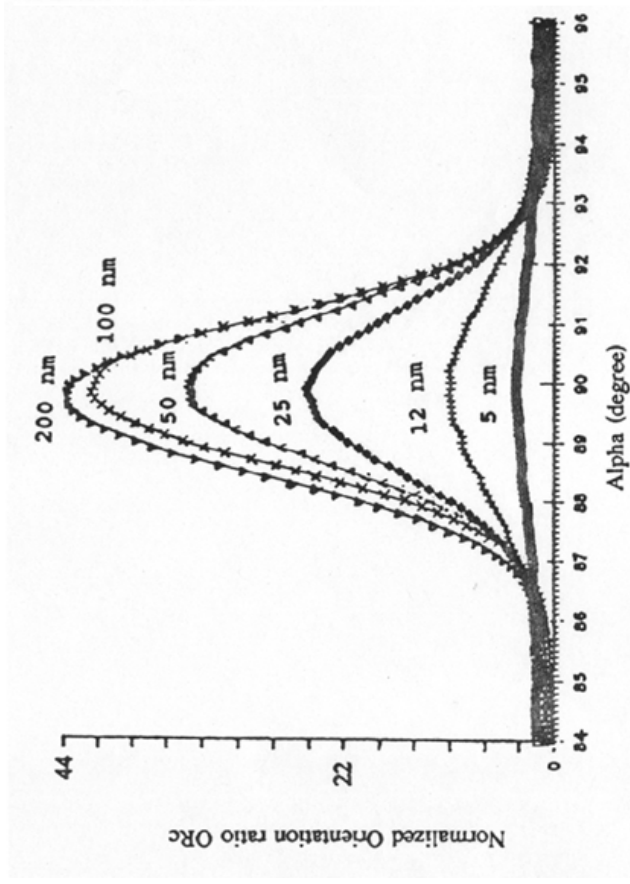
TEXTURE

**Co-Cr FILMS FOR
MAGNETIC
RECORDING MEDIA**



Results

THICKNESS: (5 - 1000) nm
SUBSTRATE: (A) Ge/SiO₂/Si
METHOD: SPUTTERING
COMPOSITION: Co₈₁Cr₁₉

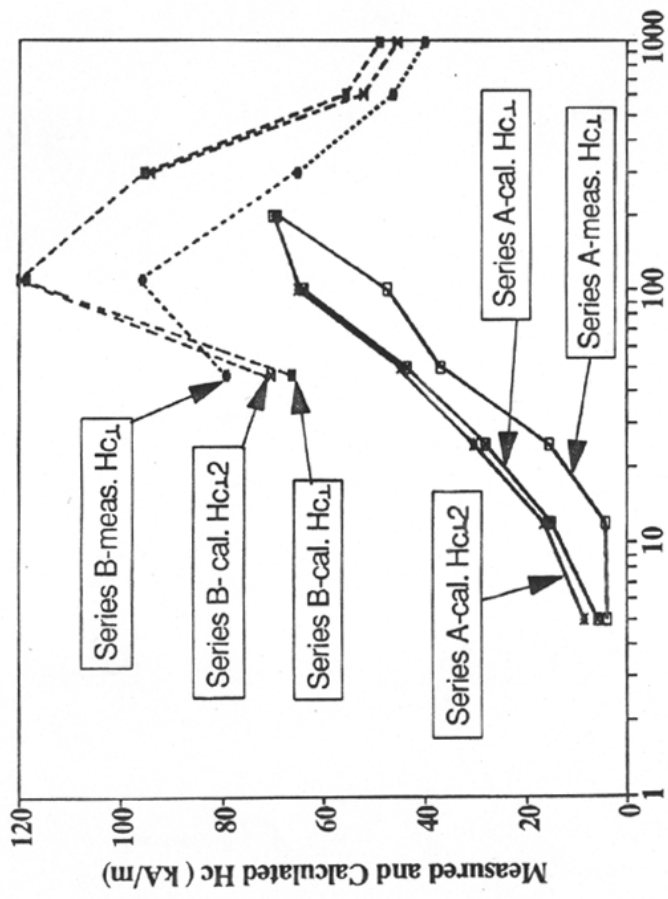
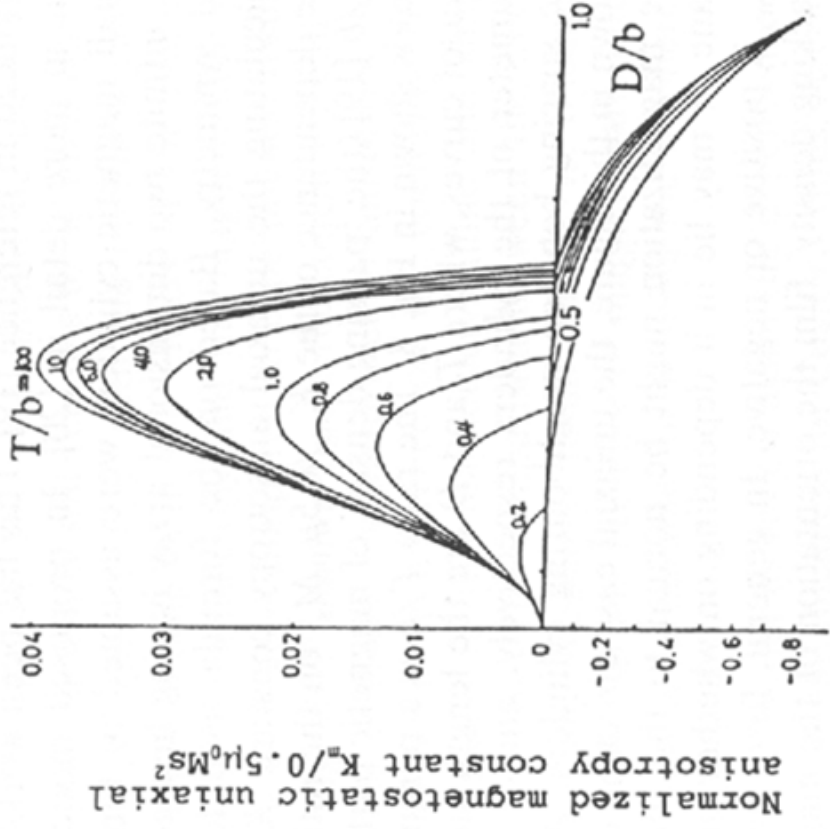


Model of Correlation Between Texture and Anisotropy of Magnetic Properties

- Uniaxial magnetic anisotropy of grains
- Unisome rotation of magnetization of particles towards the direction of the magnetic field
- Texture described by

$$P(\alpha) = \sum_{l=0}^L a_l^v k_l^v(\alpha)$$

The Measured and the Calculated $H_{c\perp}$ (With Magnetostatic Interaction Between Columnar Grains)

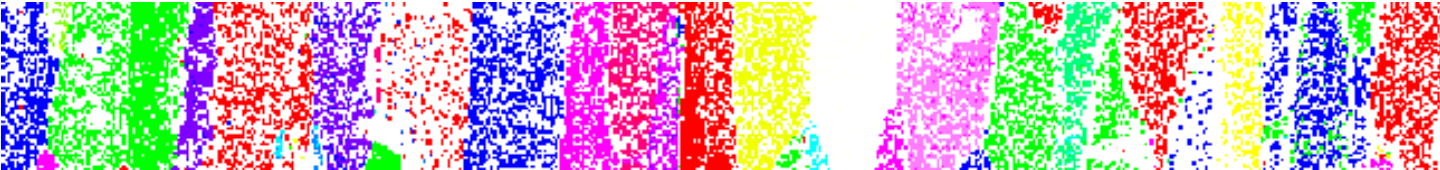


CONCLUSIONS

- TEXTURE PLAYS AN IMPORTANT ROLE IN OPTIMIZING THE PROPERTIES OF Co-Cr FILMS
- MAGNETIC PROPERTIES CAN BE CORRELATED TO ORIENTATION AND SHAPE OF GRAINS
- TEXTURE STRENGTH DEPENDS ON THE FILM THICKNESS

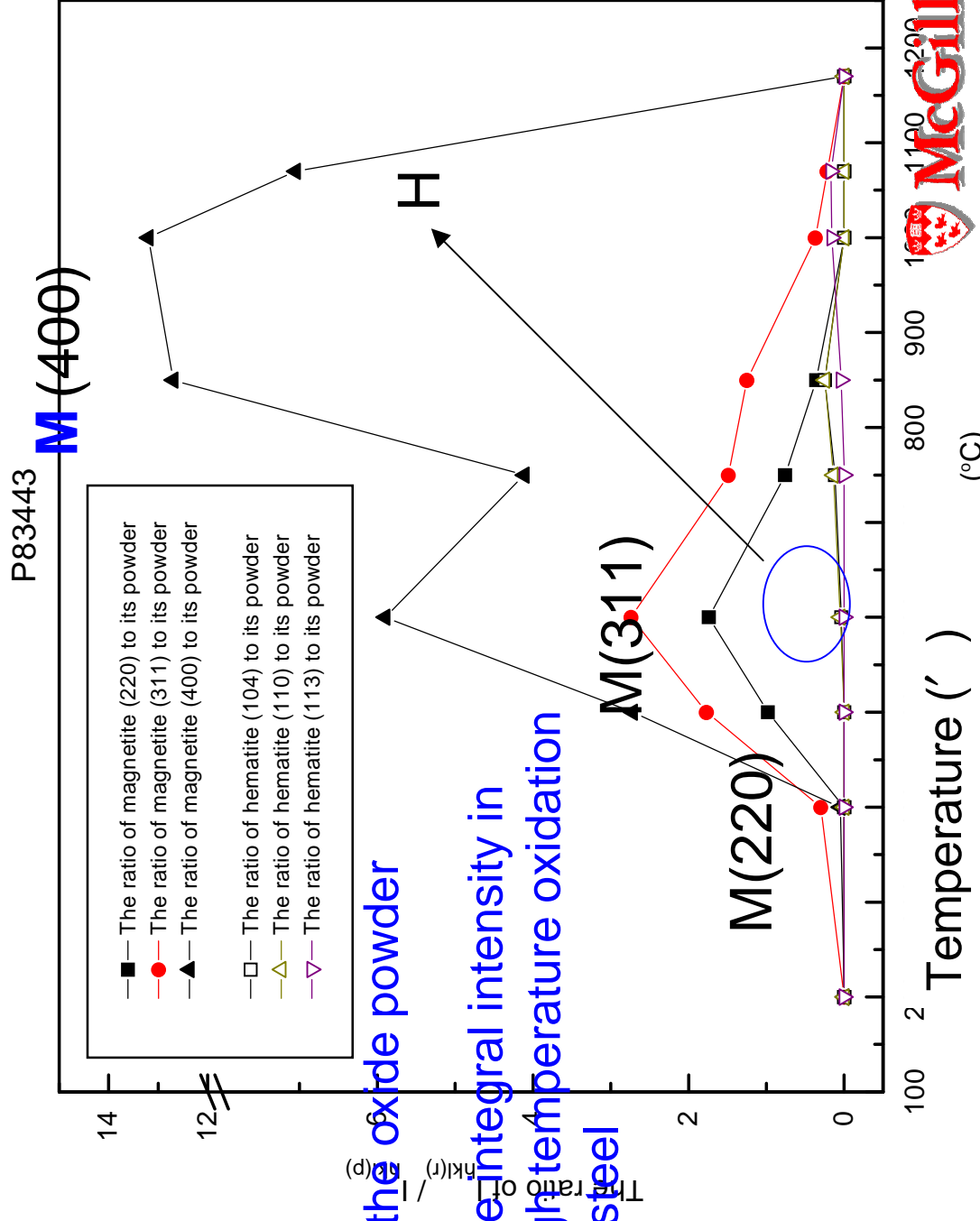


**Texture in iron
oxides**



Anisotropy of the growth of iron oxides

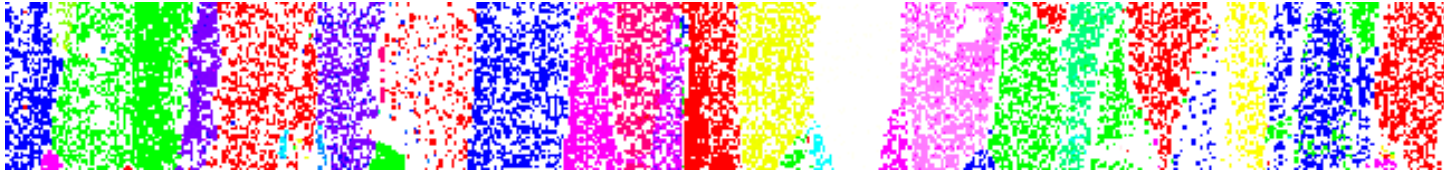
The relative intensity ratio of magnetite and hematite peaks as a function of temperature.



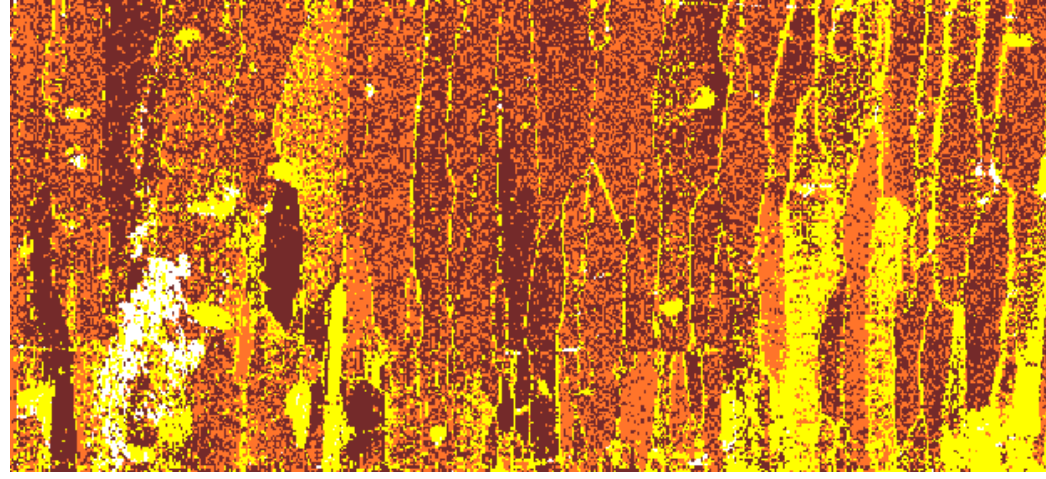
The ratio of $I_{hkl(r)} / I_{hkl(p)}$

In the oxide powder

The integral intensity in high temperature oxidation of steel



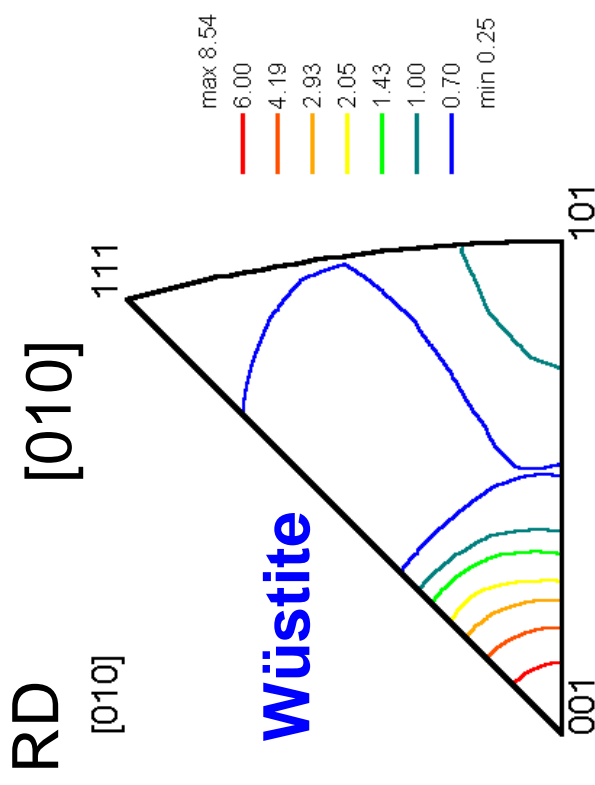
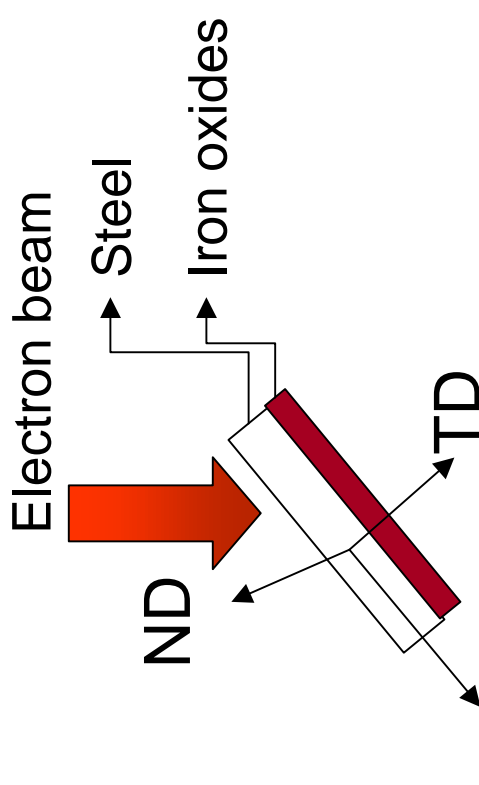
Anisotropy of oxide growth and texture



Free surface

→ Fe

OIM



Wüstitite

22.50 μm = 45 steps

CONCLUSIONS

- WUSTITE, MAGNETITE AND HEMATITE ARE TEXTURED
- TEMPERATURE OF OXIDATION PLAYS AN IMPORTANT ROLE IN CONTROLLING THE STRENGTH AND TYPE OF TEXTURE

TEXTURE
TEXTURE



ELECTROMIGRATION
FAILURE

Typical Electromigration Failures

Figure IV.1.1.(a)

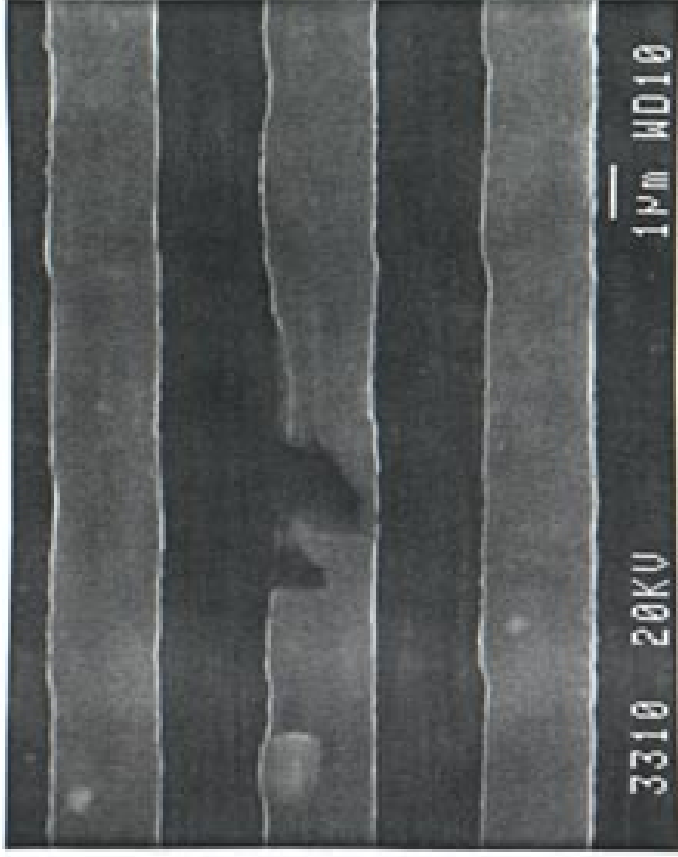
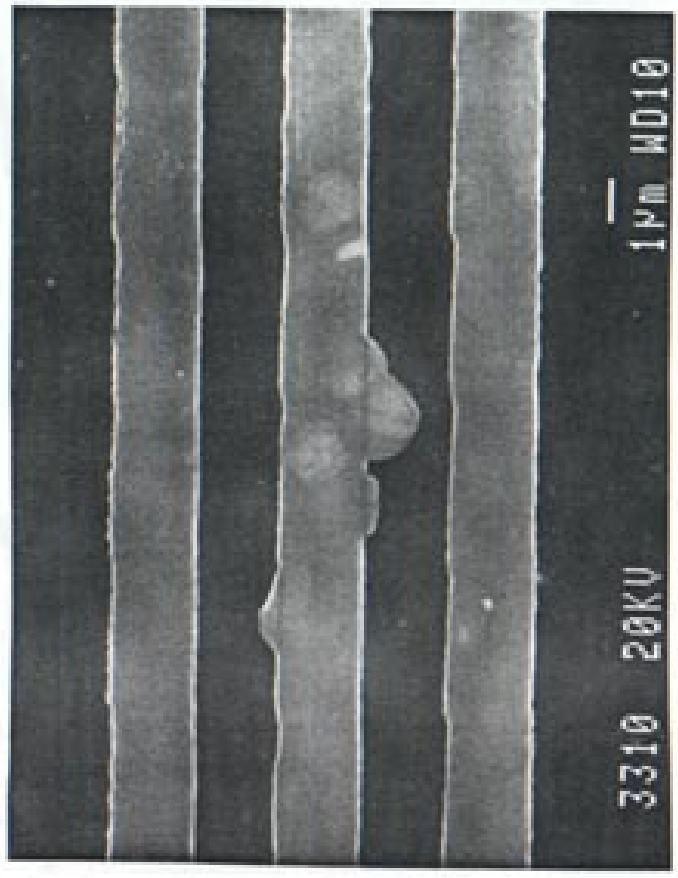
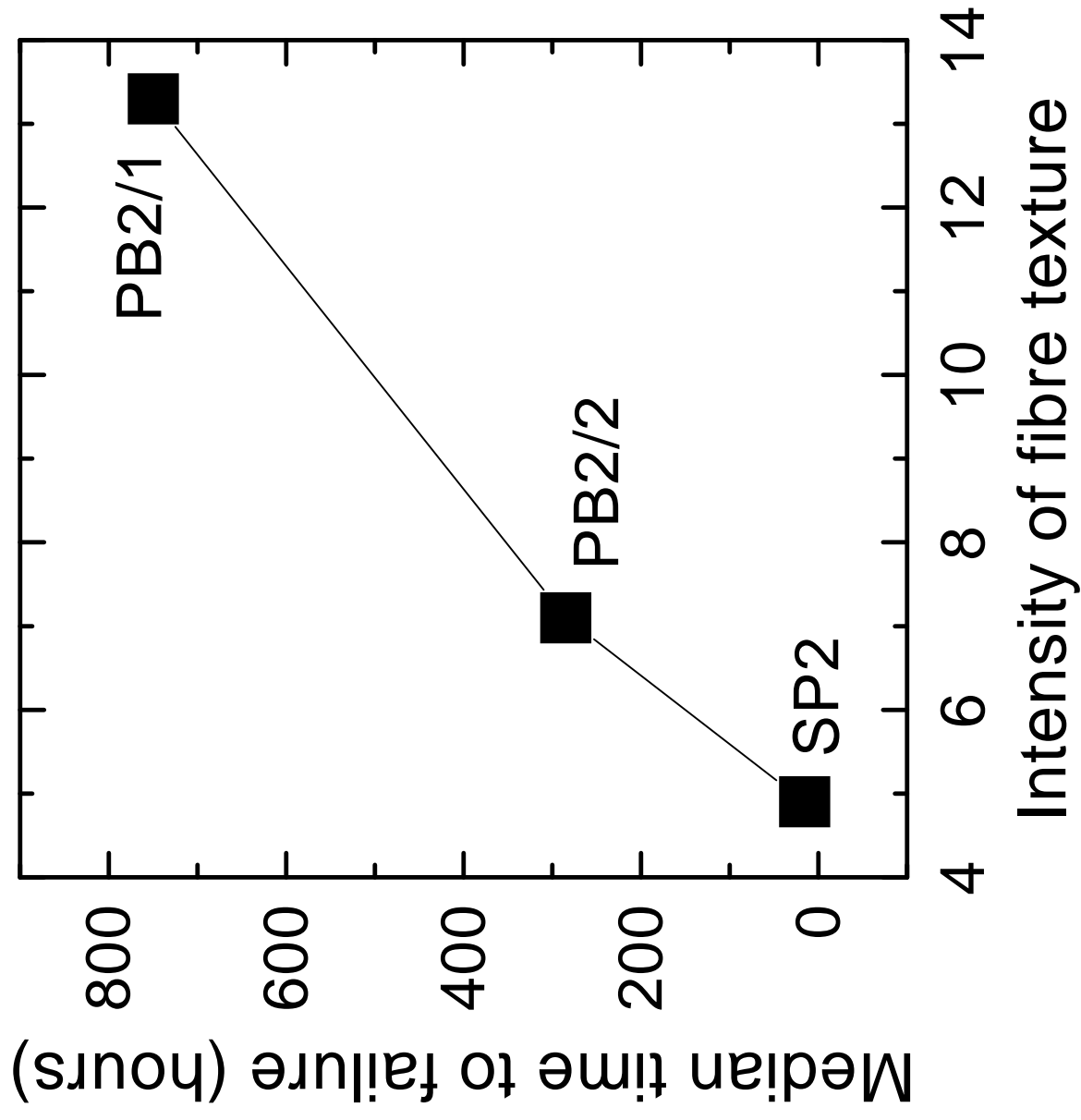
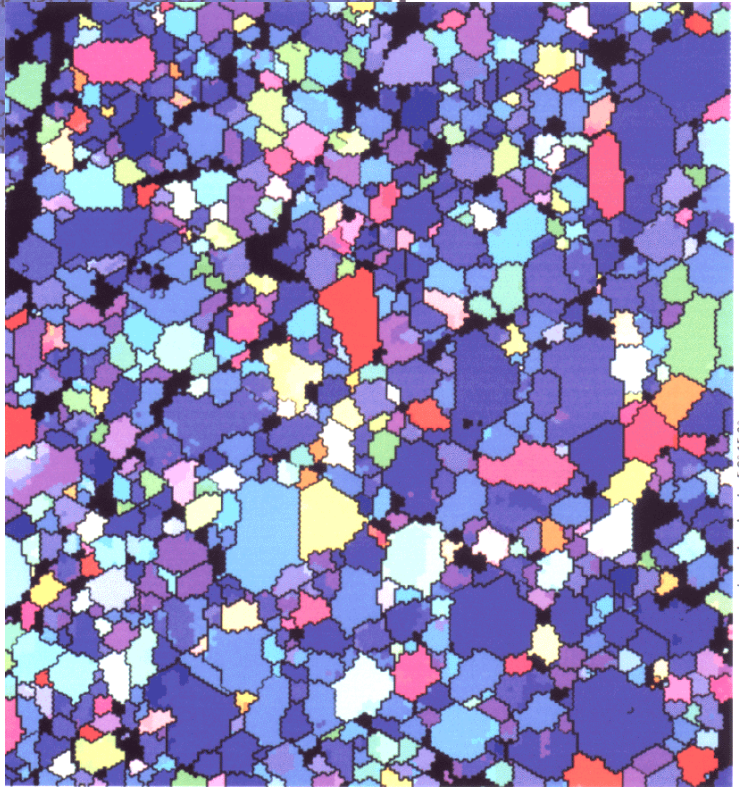
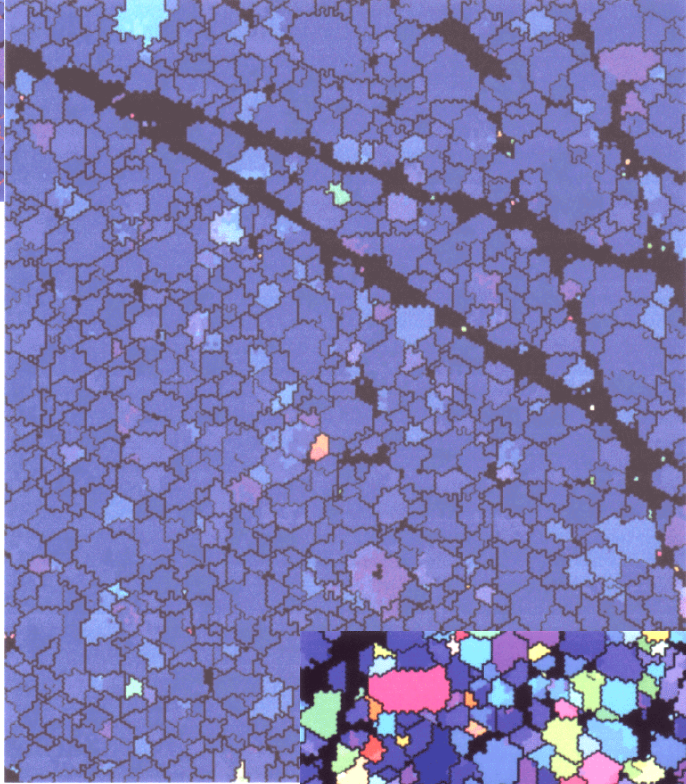
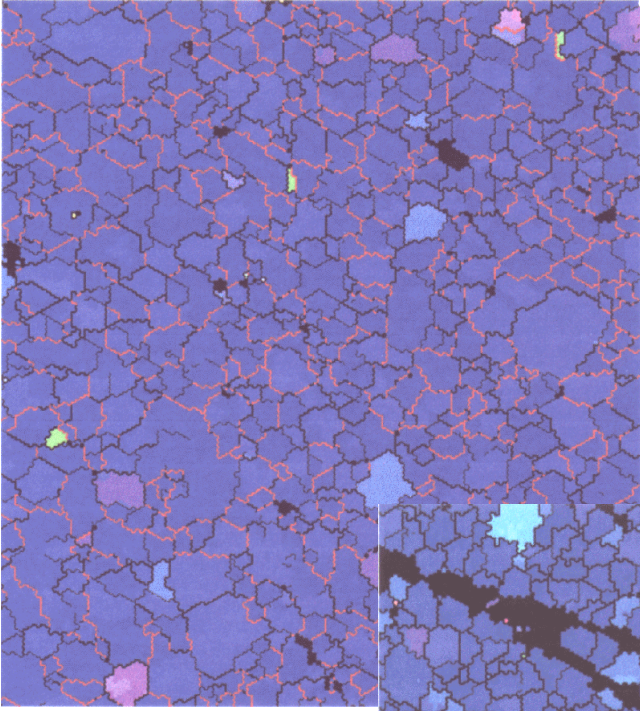
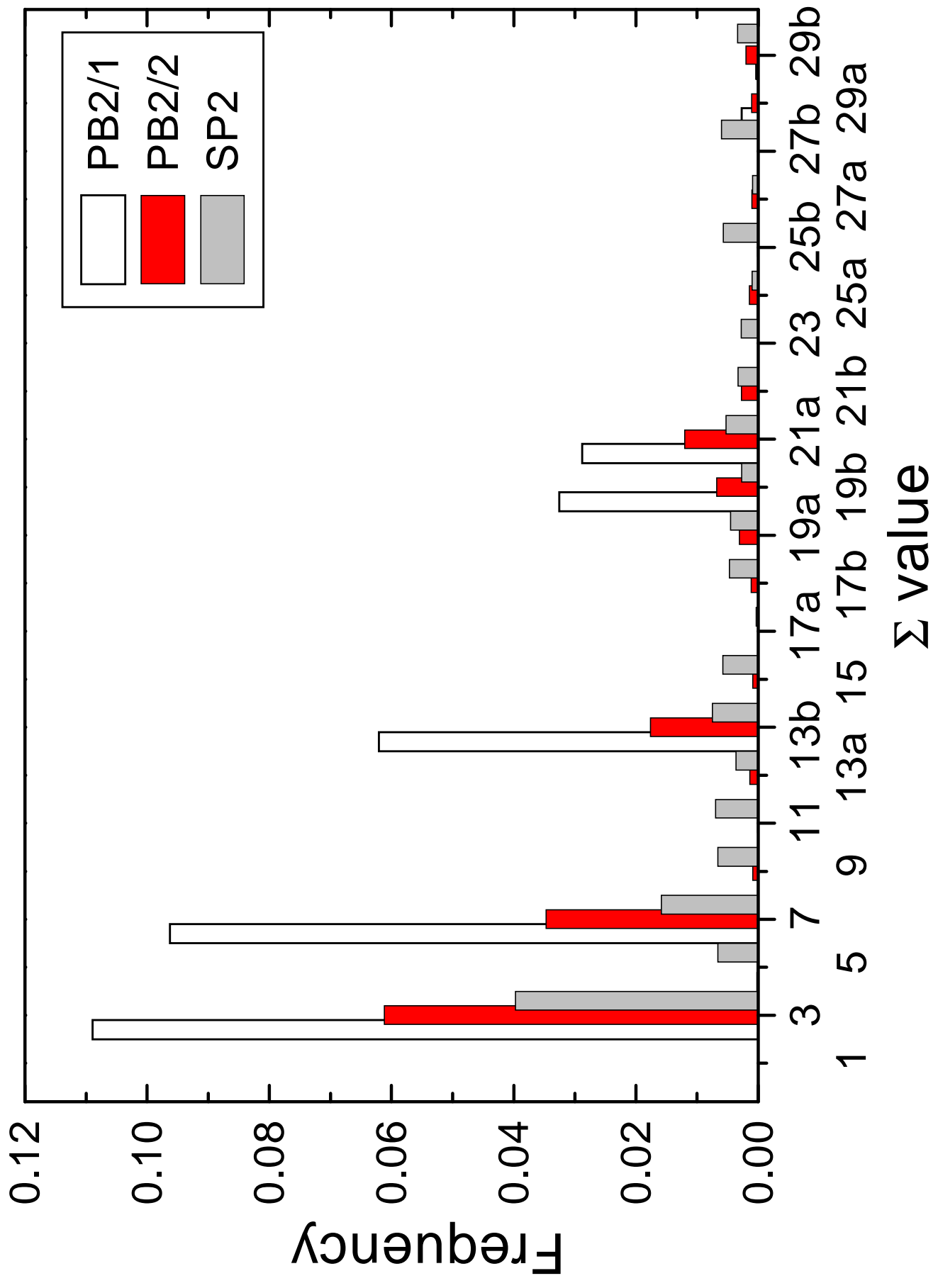


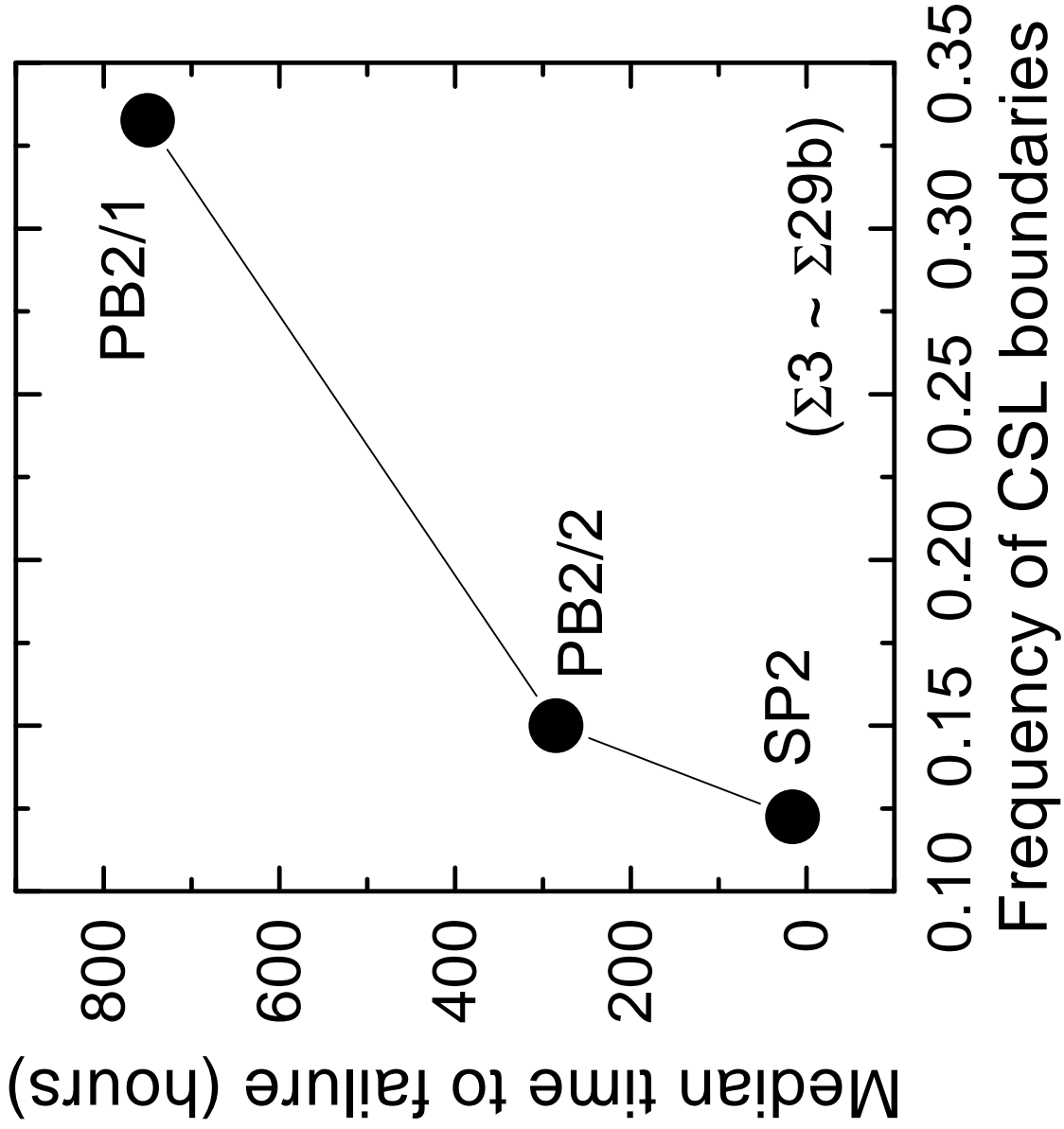
Figure IV.1.1.(b)

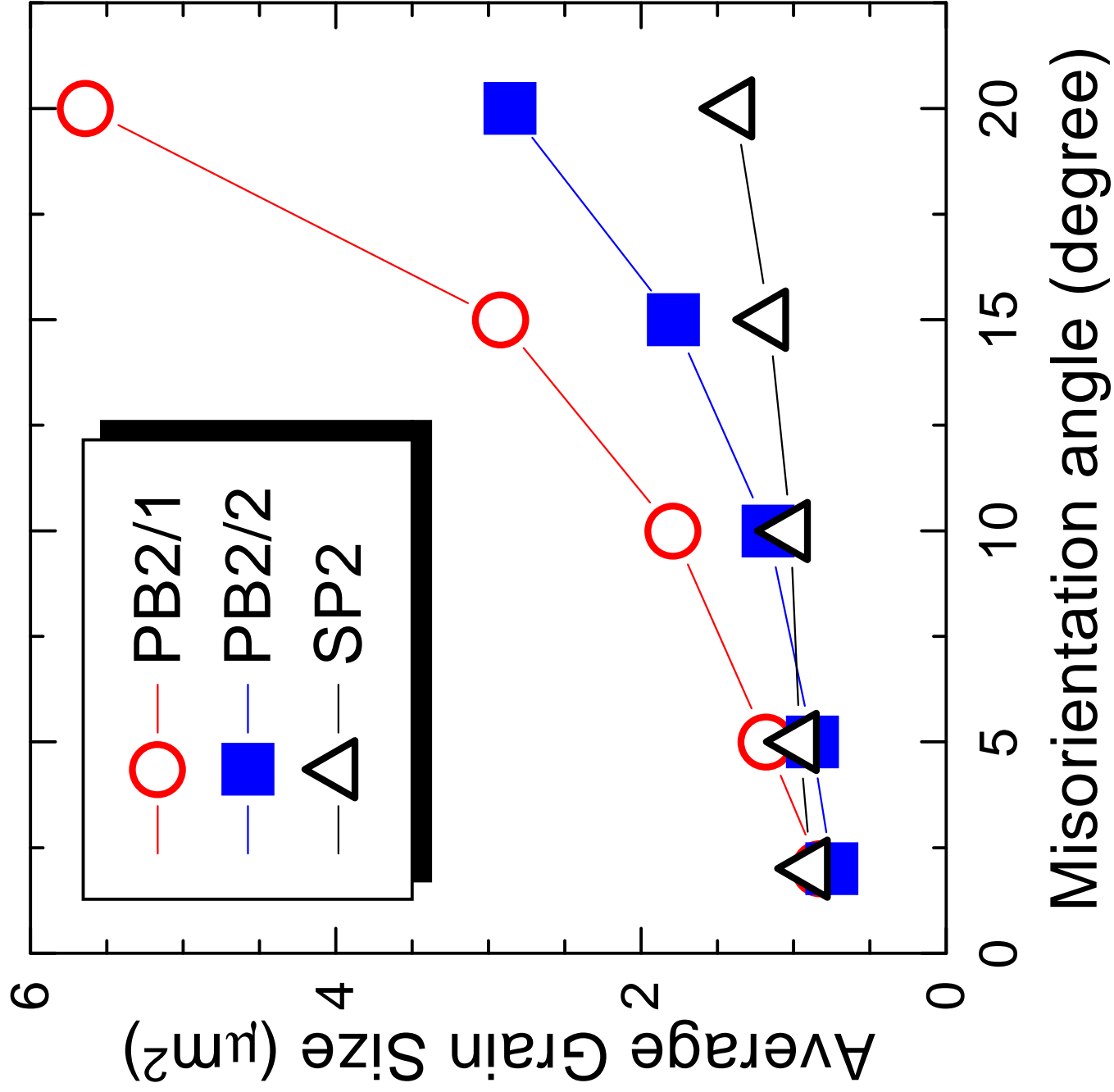












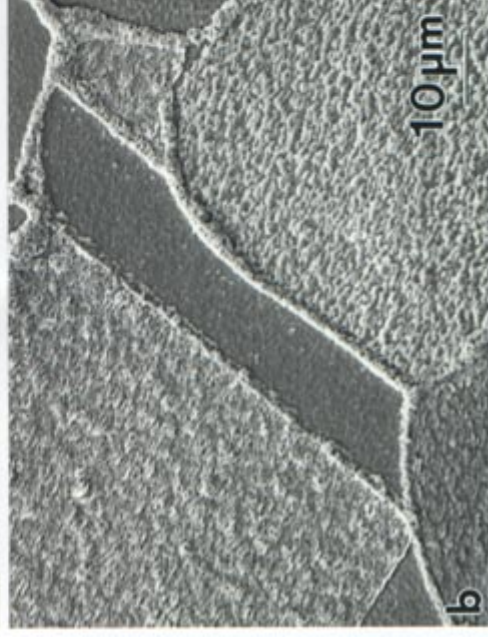
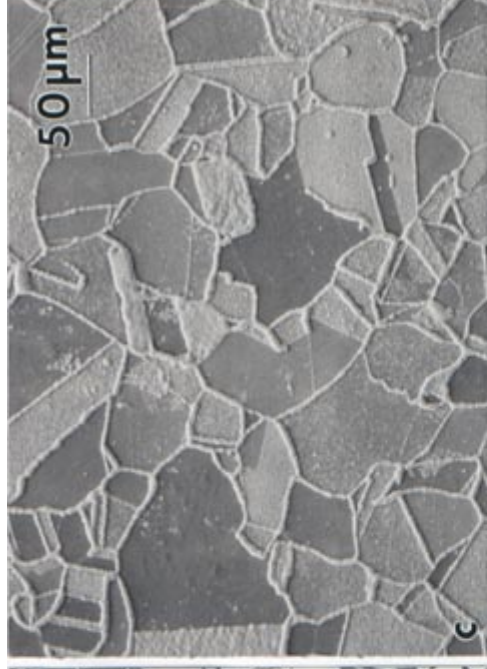
Message

- Texture engineering can be used to improve the life time of electronic interconnects.
- Better understanding of a role of texture in failure of electronic multi-layer systems and mass transport through diffusion processes is required.

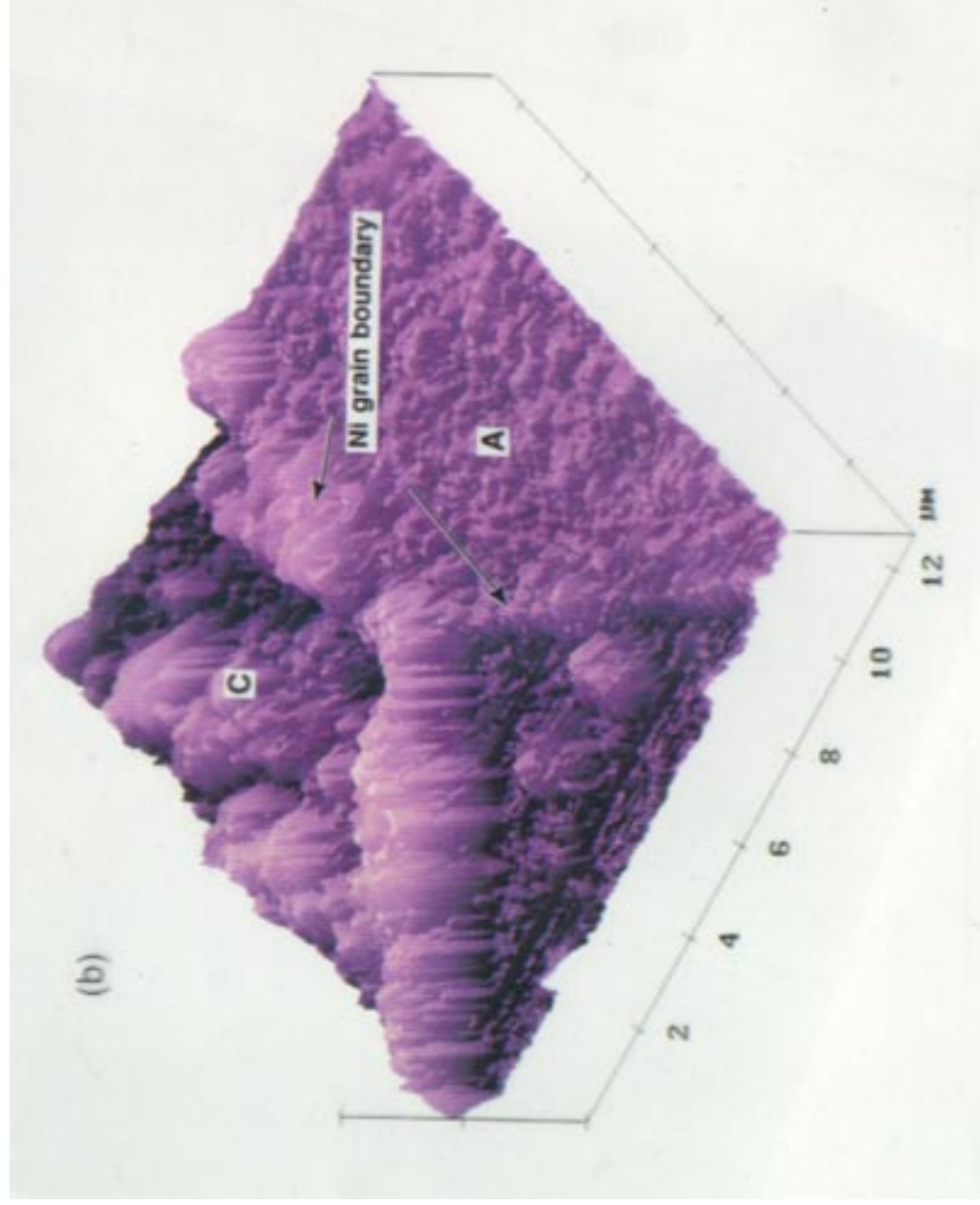
TEXTURE

HIGH
TEMPERATURE
OXIDATION

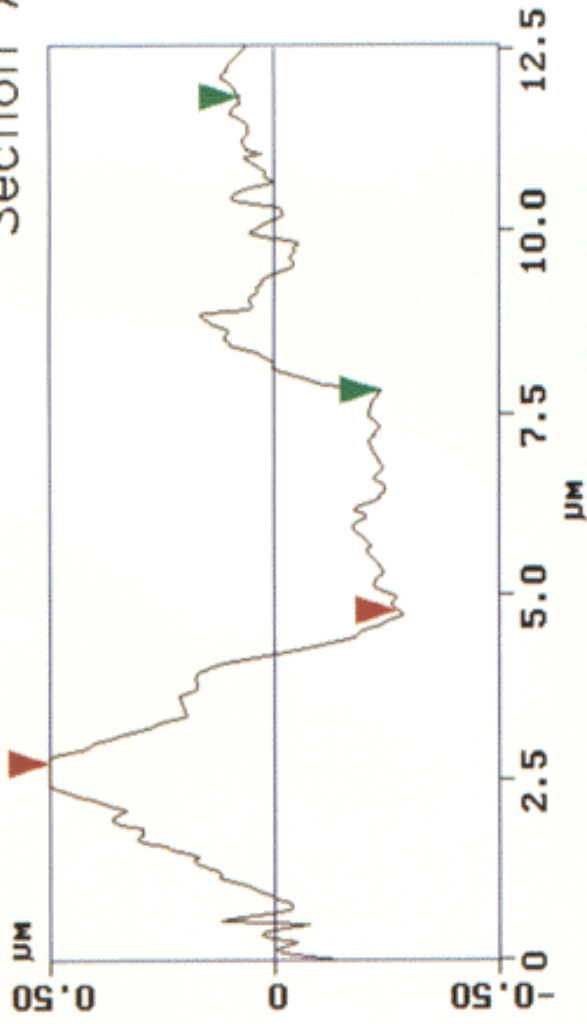
Oxidation of Polycrystalline Nickel



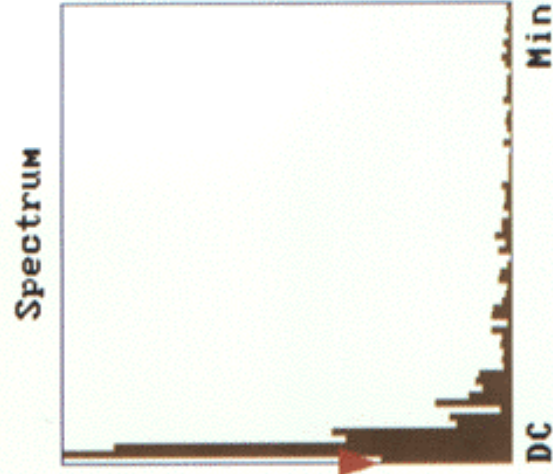
Oxidation at Grain Boundaries



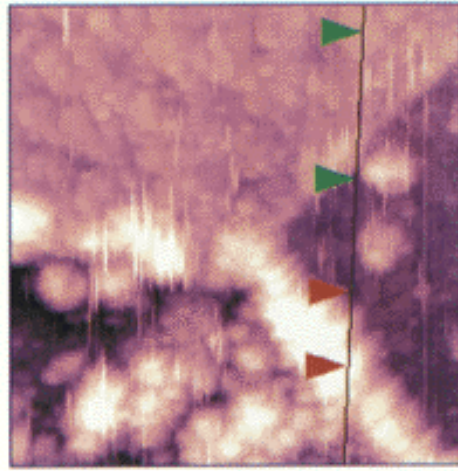
Section Analysis

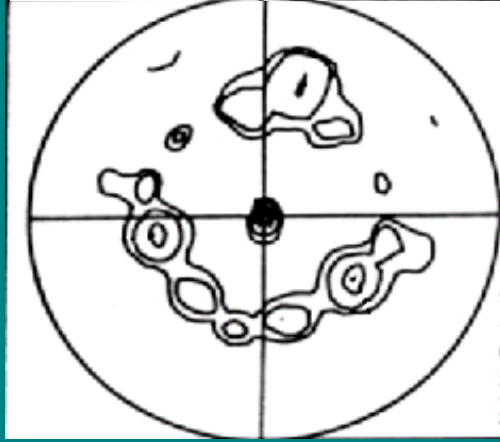


| | | |
|--------|--------|----|
| L | 3.993 | μm |
| RMS | 63.215 | nm |
| Ic | 264.65 | nm |
| Ra(Ic) | 6.874 | nm |
| Rmax | 393.16 | nm |
| Rz | 147.18 | nm |
| Rz Cnt | valid | |

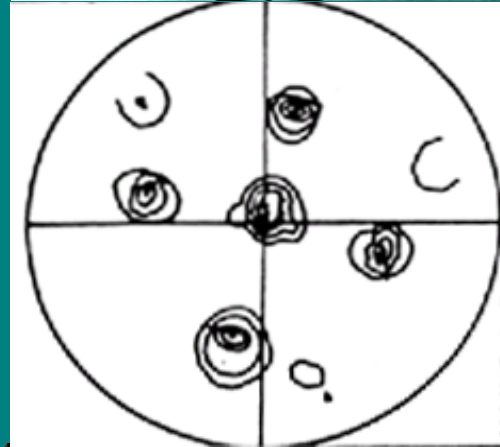


| | | |
|-------------------|--------|-----|
| Horiz distance(L) | 2.070 | μm |
| Vert distance | 806.51 | nm |
| Angle | 21.282 | deg |
| Horiz distance | 3.993 | μm |
| Vert distance | 316.31 | nm |
| Angle | 4.529 | deg |
| Horiz distance | | |
| Vert distance | | |
| Angle | | |
| Spectral period | 264.65 | nm |
| Spectral freq | 283.95 | Hz |
| Spectral amp | 2.152 | nm |

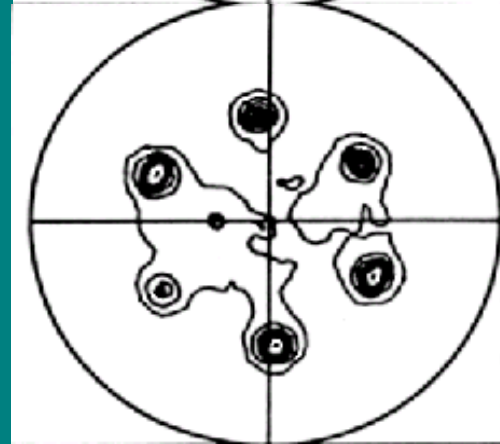




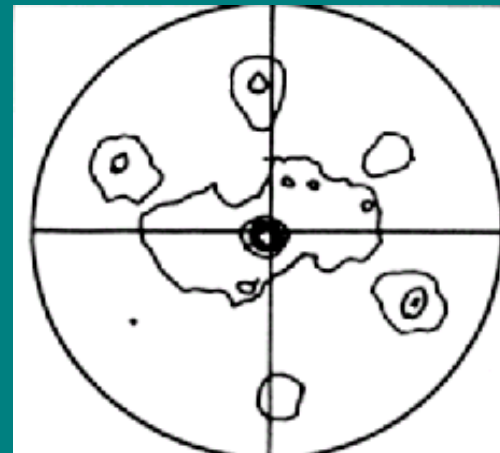
(200) Pole figure
Ident.: NiO
Contour levels: 1, 2, 4, 8, 16, 32, 64



(111) Pole figure
Ident.: NiO
Contour levels: 1, 2, 4, 8, 16, 32, 64



(200) Pole figure
Ident.: NiO
Contour levels: 1, 2, 3, 4, 5, 6, 7

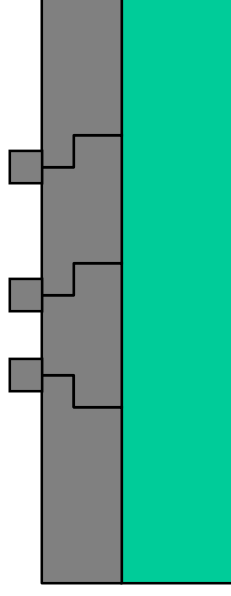


(111) Pole figure
Ident.: NiO
Contour levels: 1, 3, 5, 7, 9, 11, 13

The proposed oxidation mechanism on Ni

- Nucleation occurs on the sample surface randomly, the orientations of the oxide nuclei are determined by the **lattice matching** between the oxide and the metal
- In the stage of oxide growth, the oxide grains with lower **surface energy** grow faster
- The re-nucleation occurs at the oxide **grain boundary** regions and the nucleation probability of an orientation is determined by its **surface energy**

$$p = \exp\left(-\beta \frac{E_{hkl}}{kT}\right)$$

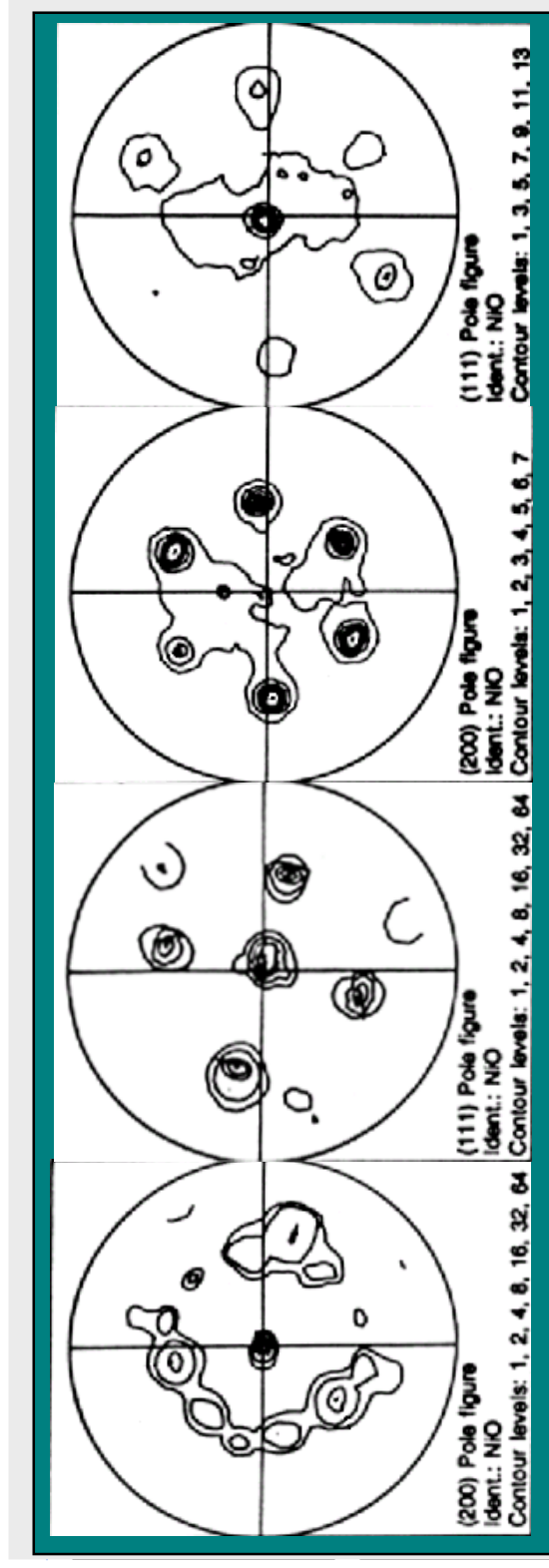
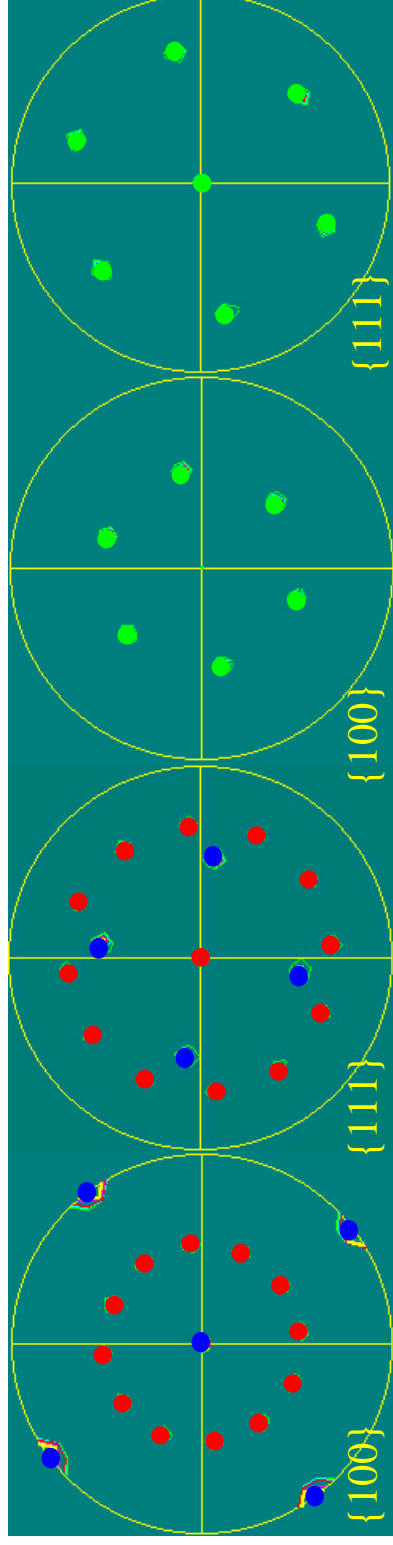


Application in Ni - NiO system

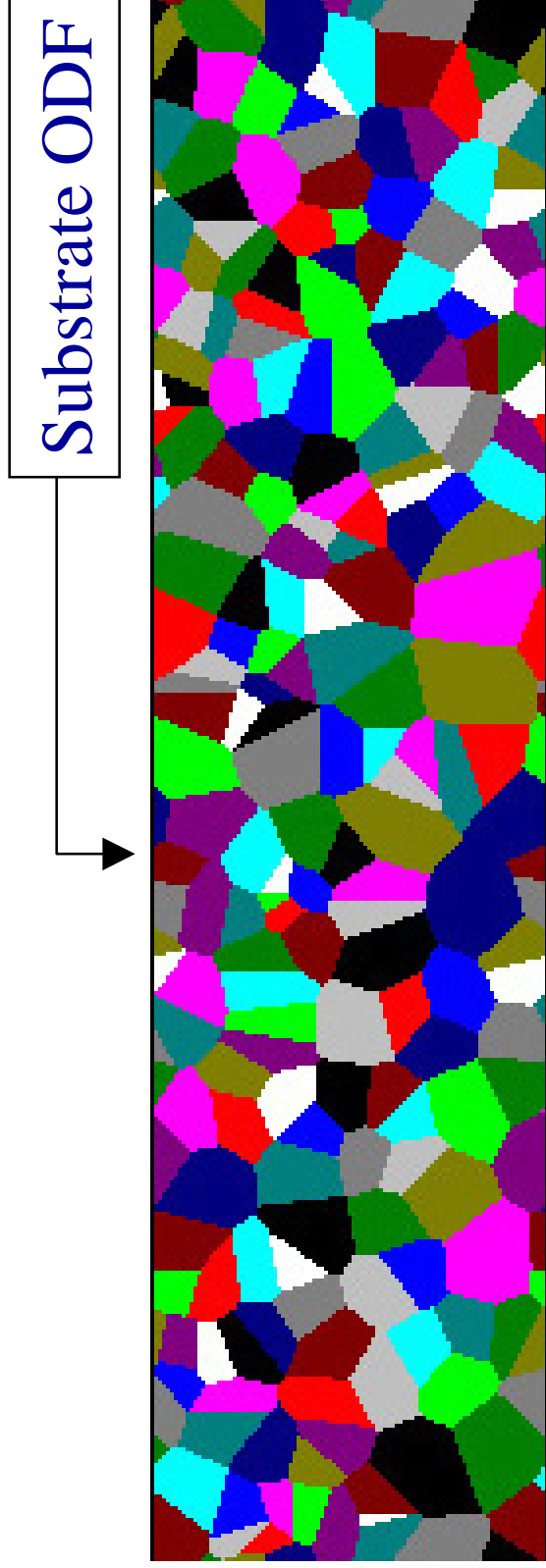
Early stage

$\{100\}_{\text{Ni}}$

$\{111\}_{\text{Ni}}$



Computer Models

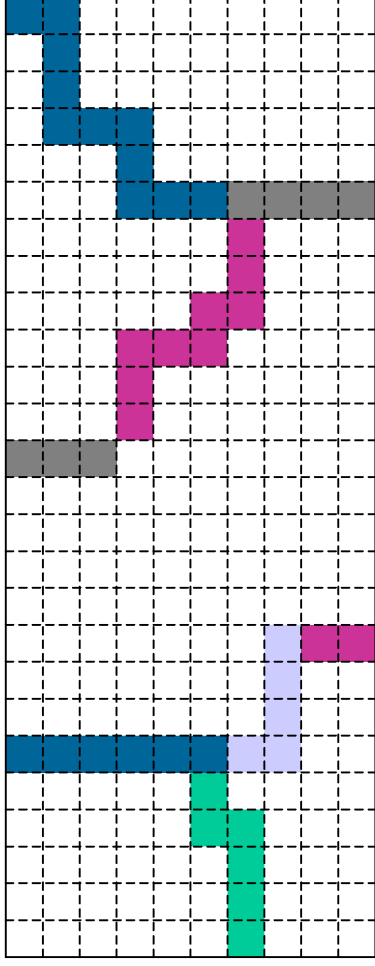


Substrate microstructure

- Substrate microstructure and ODF
- oxidation mechanism

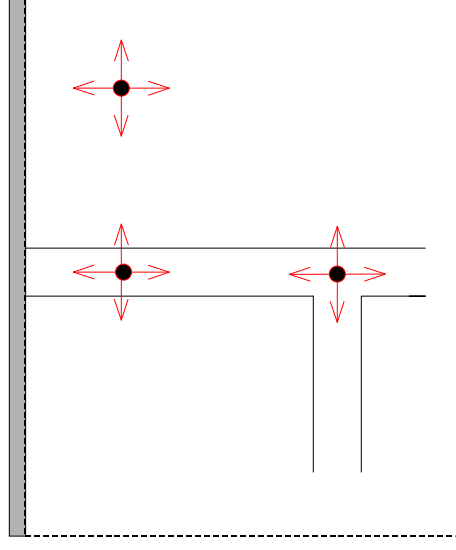
Computer Models

Oxide microstructure



Random walk: $\Gamma = nD(x, y) / \Delta x^2$

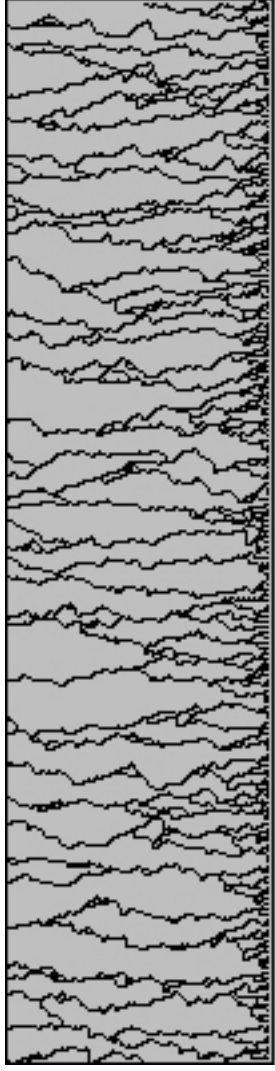
Interface



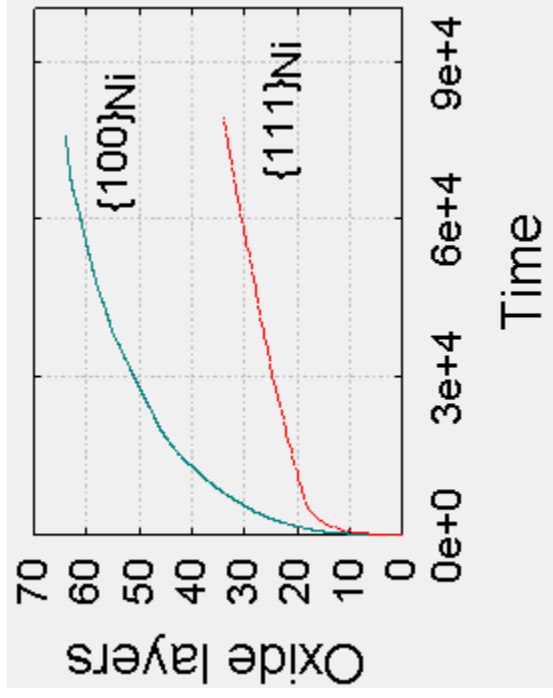
- Oxide microstructure and texture
(Grain boundary character distribution)
- Diffusion coefficient as a function of grain boundary character

Application in Ni - NiO system

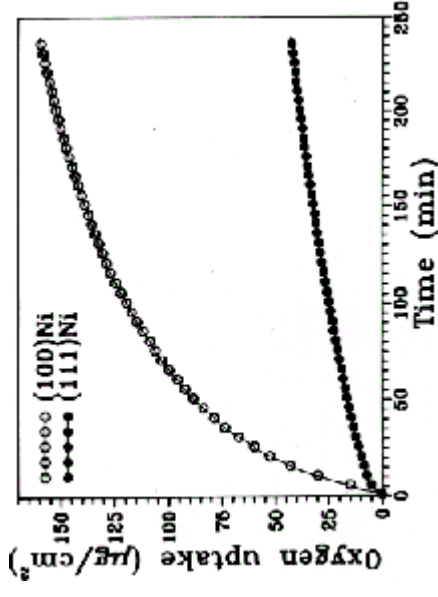
Effects of oxide texture and microstructure on the oxidation kinetics



$$D_{\Sigma 3} = D_{\text{bulk}} = 5.5 \times 10^{-15} \text{ cm}^2/\text{s}$$
$$53\% \Sigma 3 \quad D_{\text{gb}} = 7.5 \times 10^{-10} \text{ cm}^2/\text{s}$$



$$K_{100}:K_{111}=10:1$$



$$K_{100}:K_{111}=14:1$$

Message

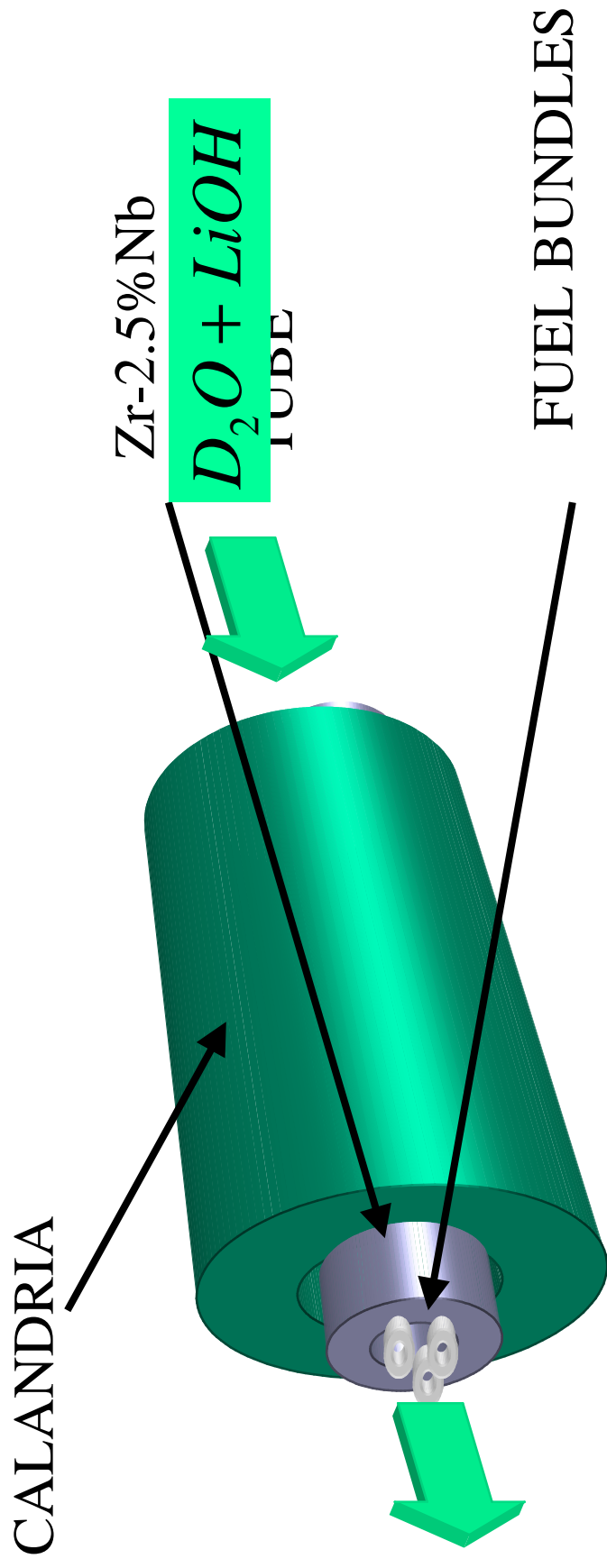
- Control of surface texture can be used to slow down high temperature oxidation in Nickel based and other alloys
- Further research is needed to understand the process of oxide texturing and diffusion of oxygen and metal through the oxide.

TEXTURE
TEXTURE

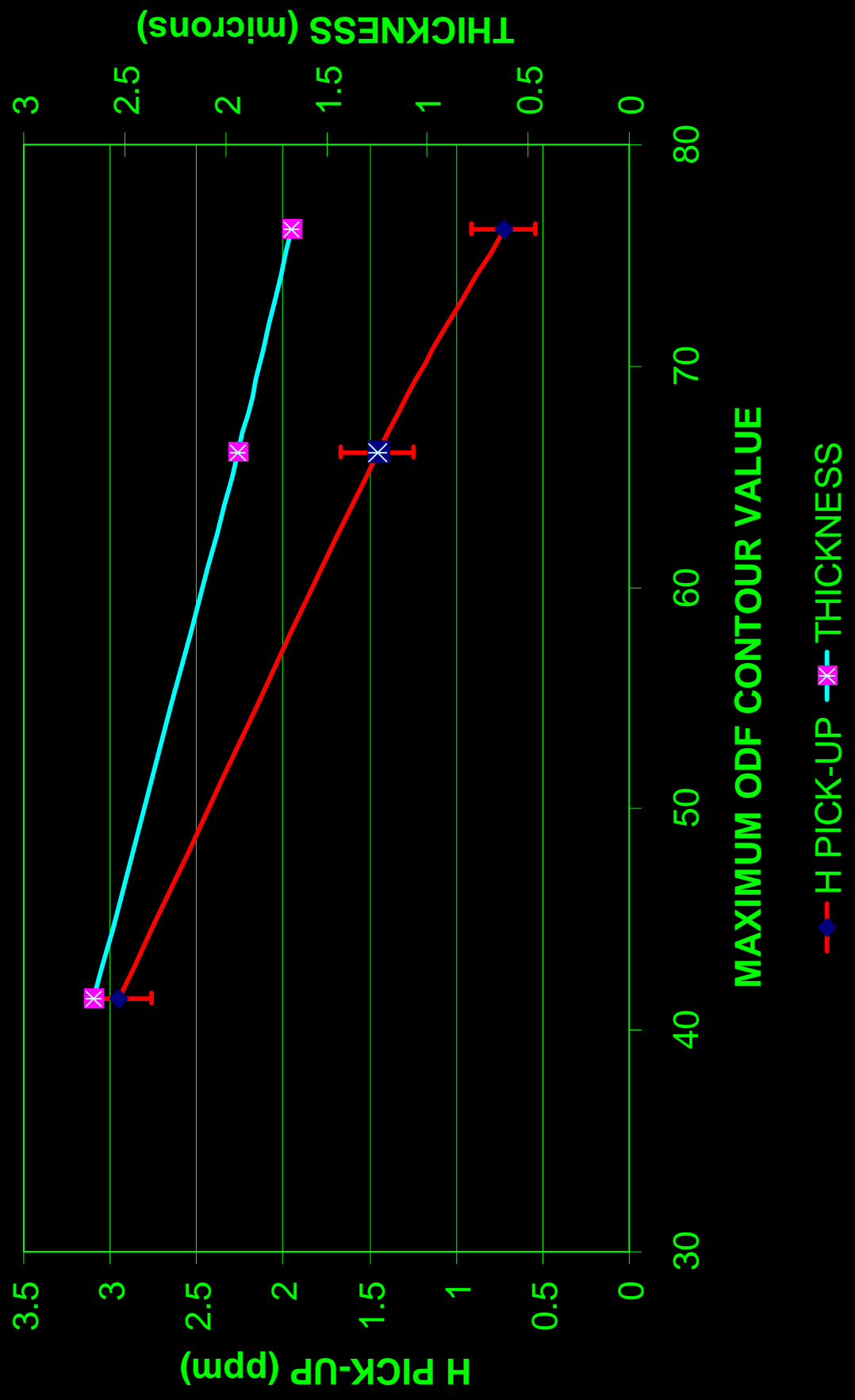


HYDROGEN INGRESS
THROUGH ZIRCONIA
FILMS

OPERATIONAL ENVIRONMENT OF THE Zr-2.5%Nb PRESSURE TUBE



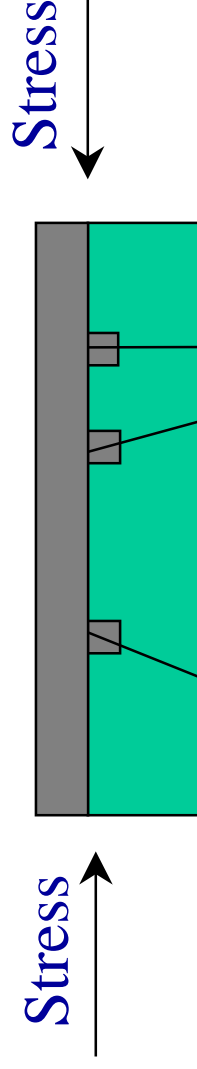
TD//SN SPECIMENS



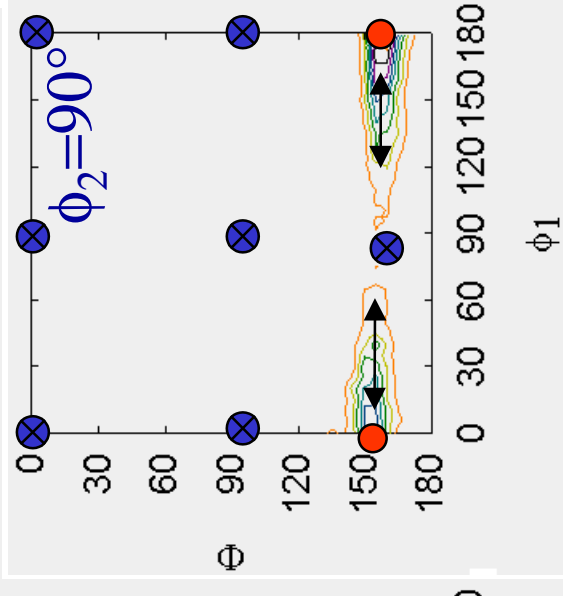
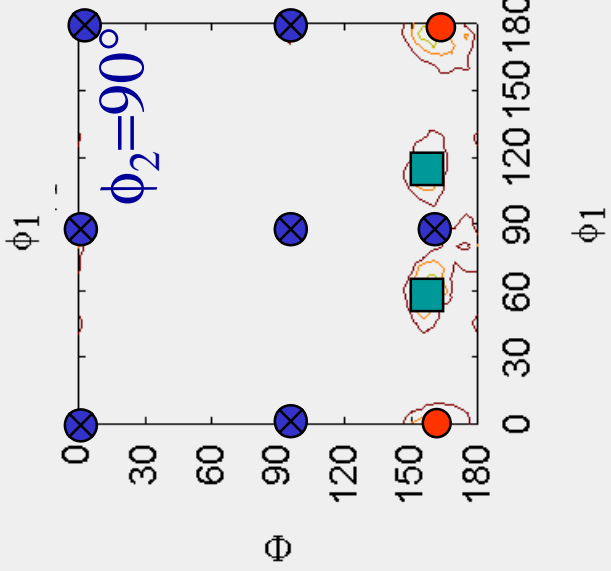
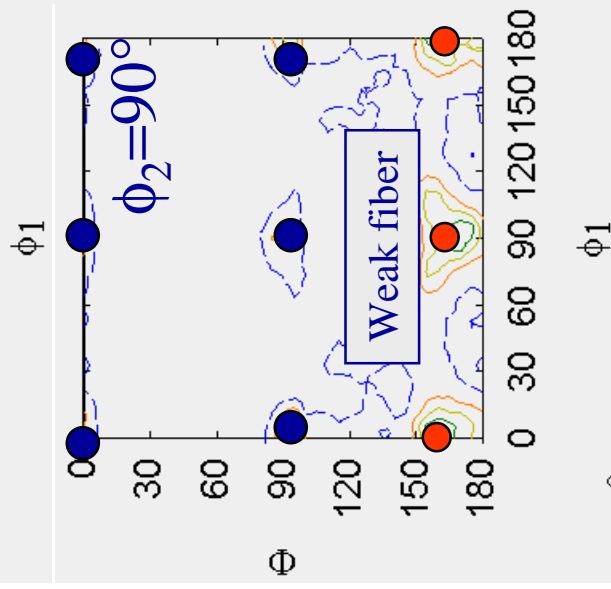
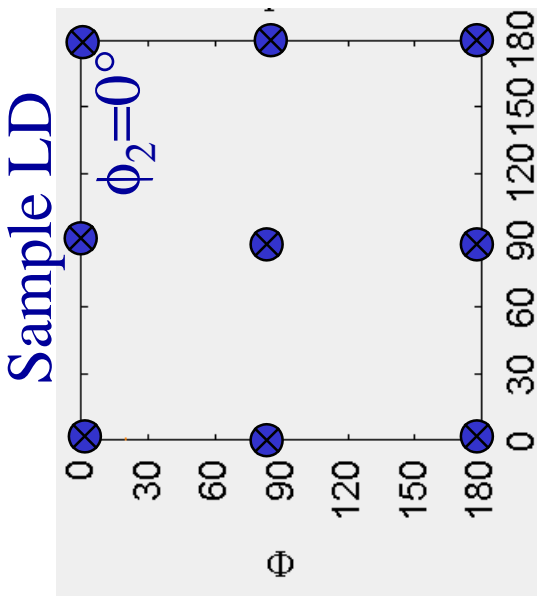
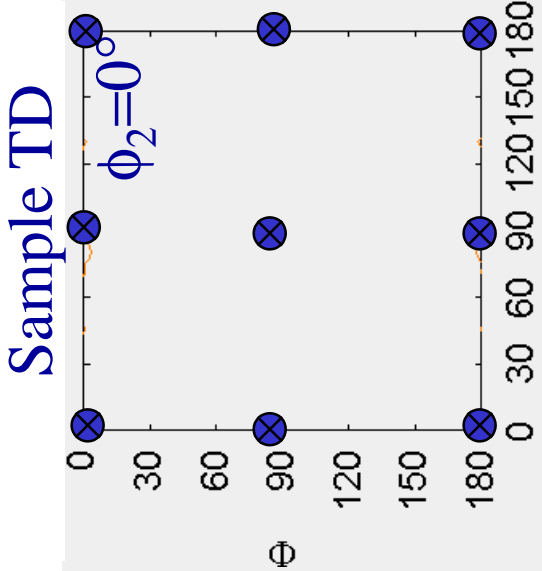
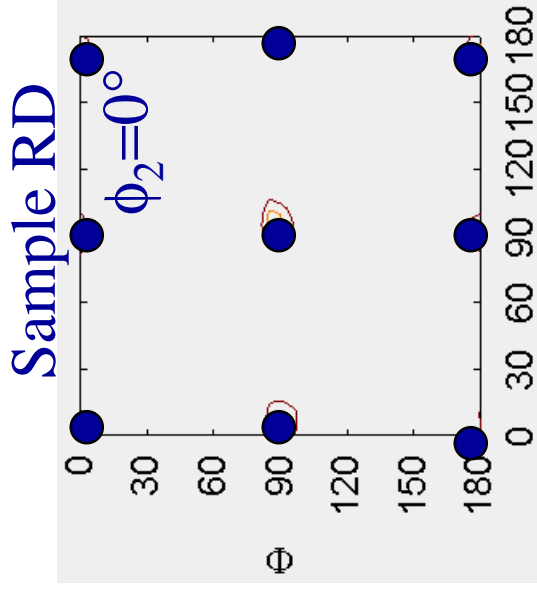
[Application in Zr - ZrO₂ system](#)

The proposed mechanism of Zr-2.5%Nb oxidation

- Nucleation occurs on the sample surface randomly, the orientations of the oxide nuclei are determined by the **lattice matching** between the oxide and the metal
- In the stage of oxide growth, **minimization of the stress**, parallel to the sample surface, plays a main role.
- The re-nucleation occurs at the **α -Zr grain boundary and β -Zr regions** and the oxide orientation is determined by **lattice matching**



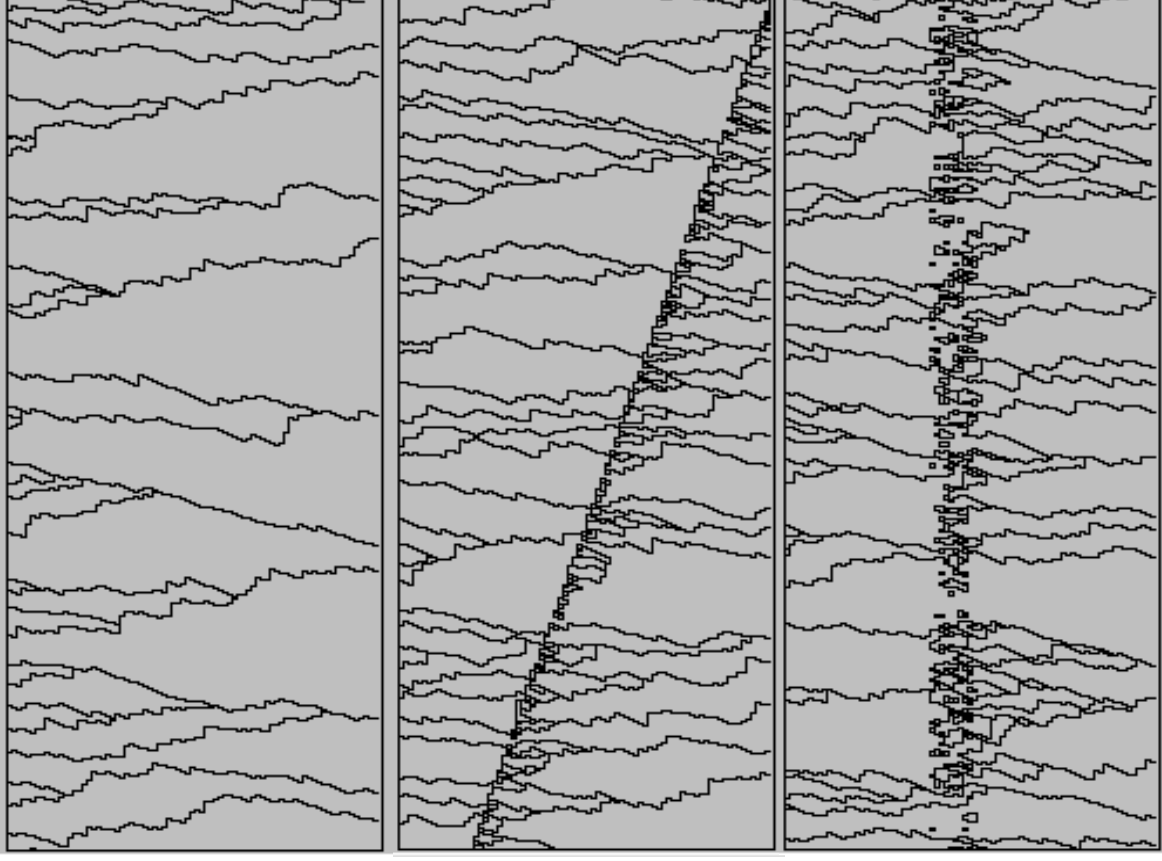
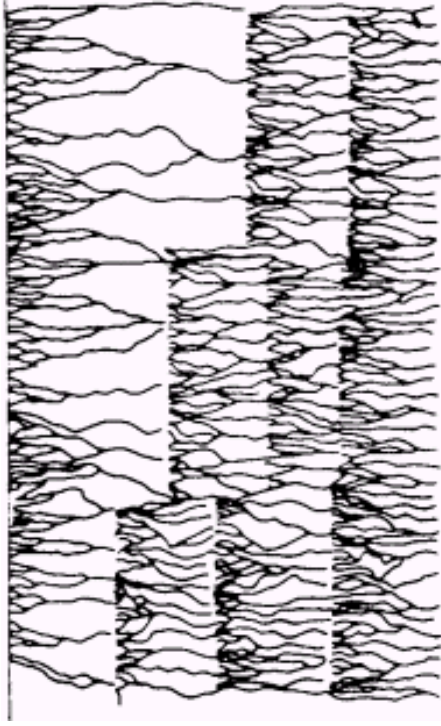
Application in Zr - ZrO₂ system



Application in Zr - ZrO₂ system

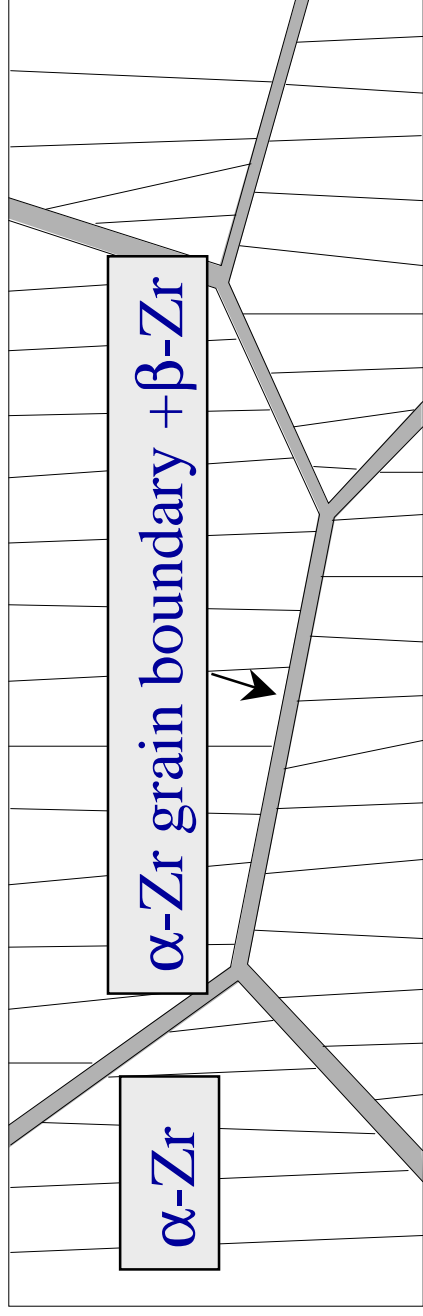
Oxide microstructure

- In the bulk of α -Zr grain
- At the α -Zr grain boundary
- In β -Zr region



Application in Zr - ZrO₂ system

Diffusion mechanism in ZrO₂

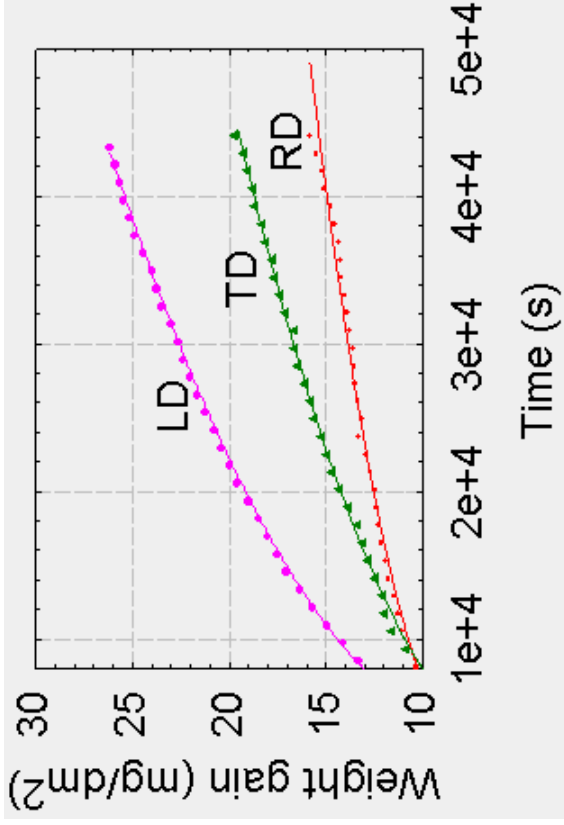


— The first diffusion path

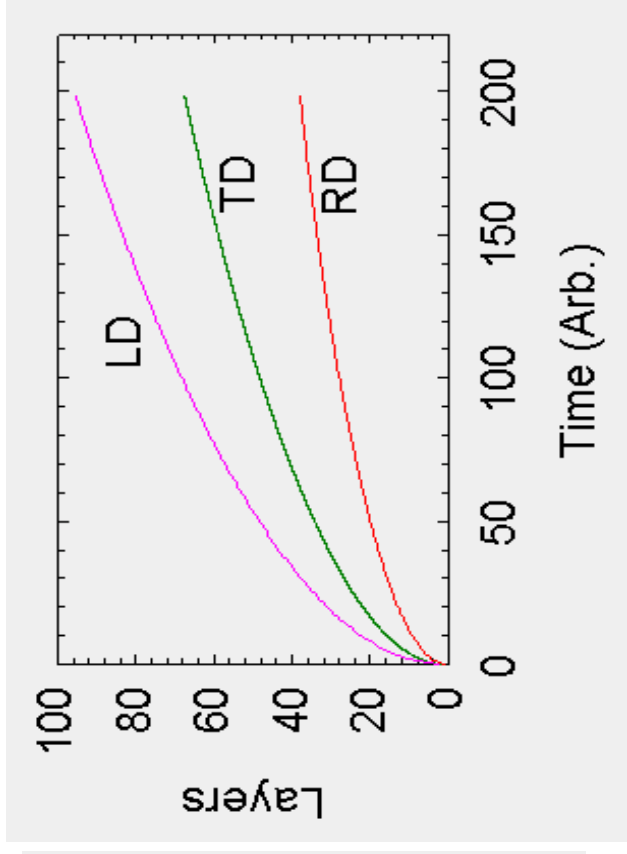
— The second diffusion path

Application in Zr - ZrO₂ system

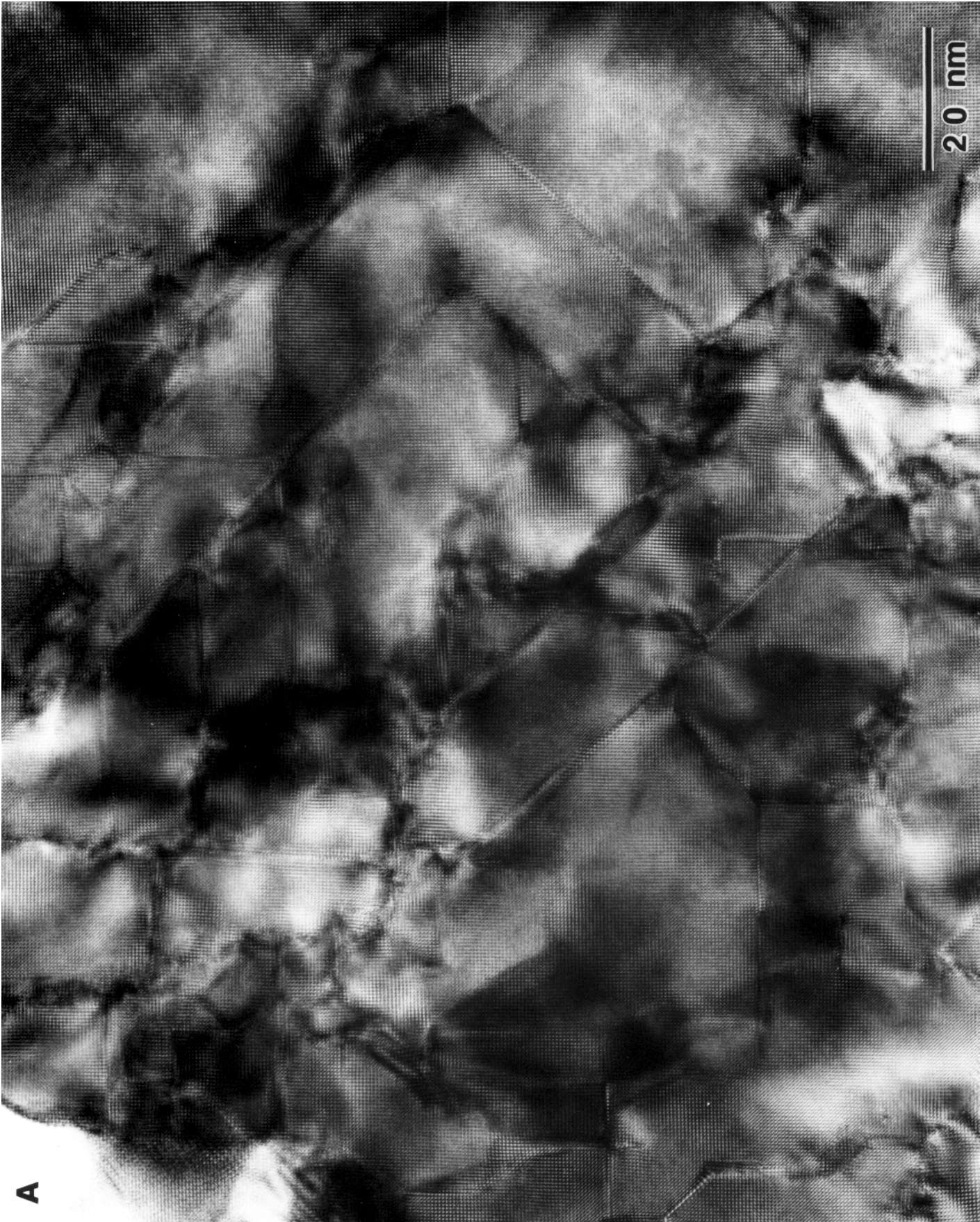
Simulation of oxidation kinetics



$$K_{RD}:K_{TD}:K_{LD}=1:3:6$$

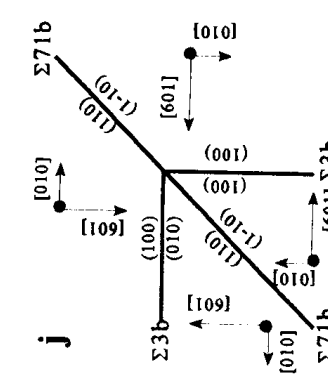
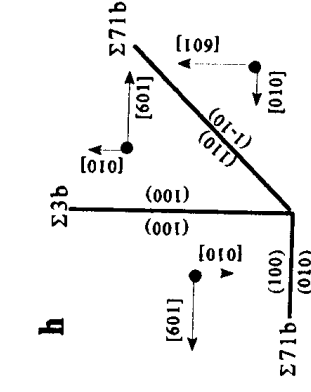
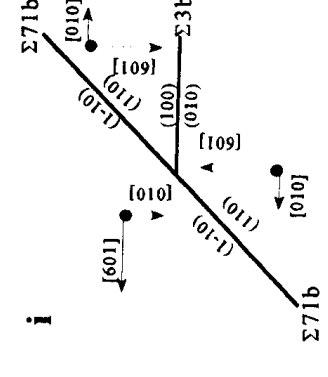
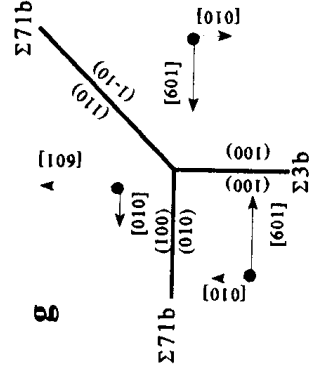
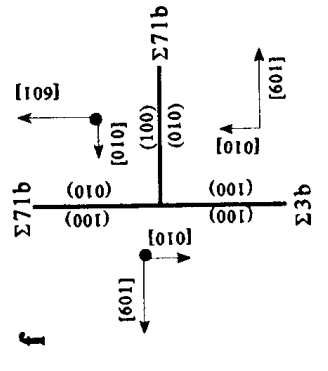
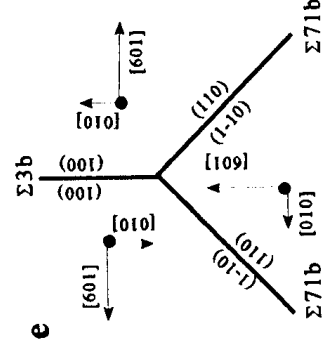
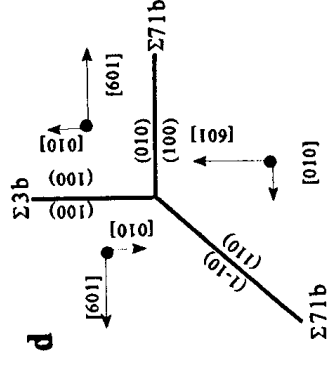
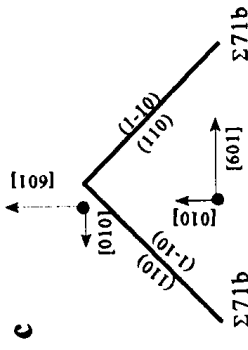
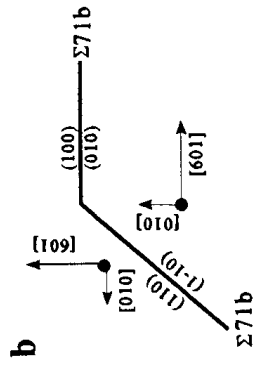
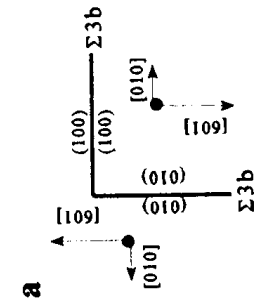


$$K_{RD}:K_{TD}:K_{LD}=1:2.8:5.6$$

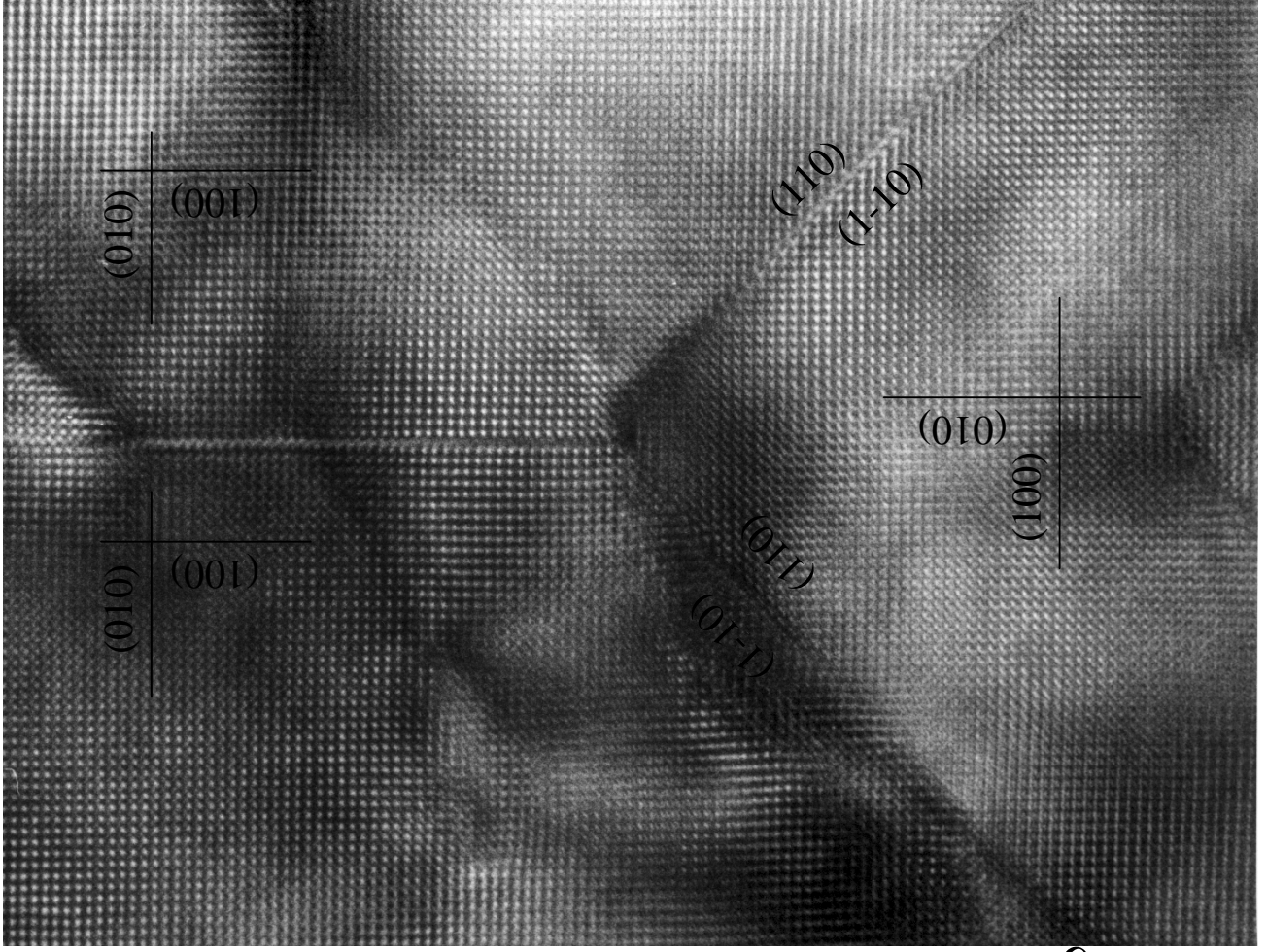




Typical Interfacial Configurations in Zirconia Film

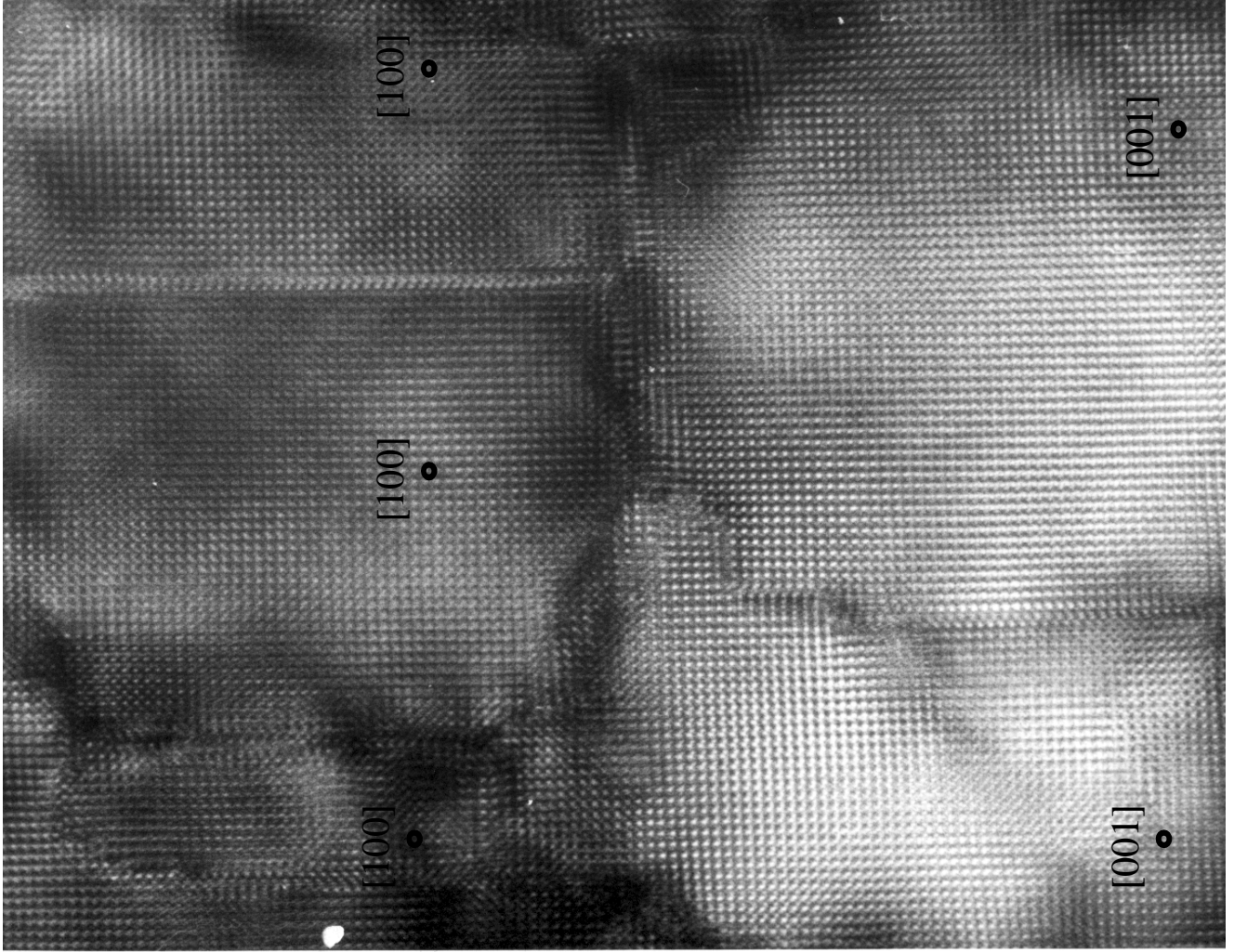


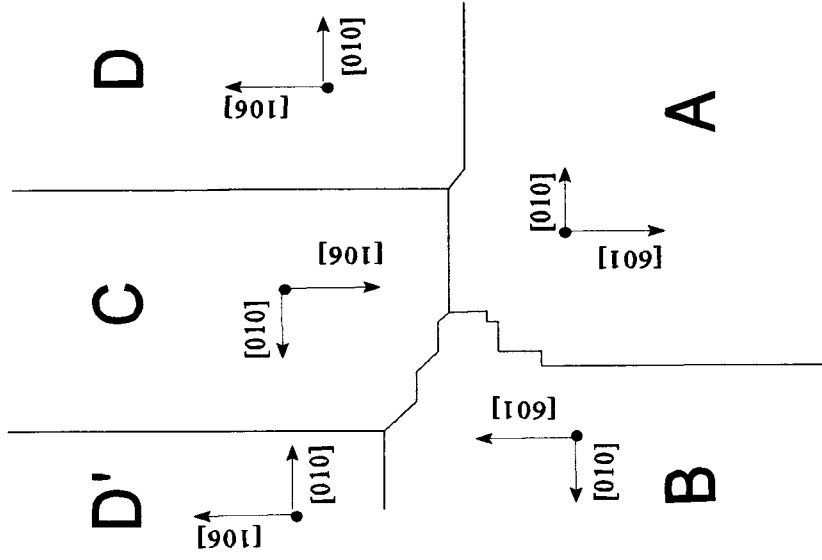
$\Sigma 3b$



$\Sigma 71b$

$\Sigma 71b$





Misorientations:

A/B $\Sigma 3b: 180^\circ[001]=180^\circ[601]$

C/D and C/D' $\Sigma 3a: 180^\circ[100]=180^\circ[106]$

A/C and B/D' $\Sigma 1': 180^\circ[101]=180^\circ[10\bar{1}]$

A/D and B/C $\Sigma 3c: 80.4^\circ[010]=99.6^\circ[010]$
 (rotation by the β angle about $[010]$)

Message

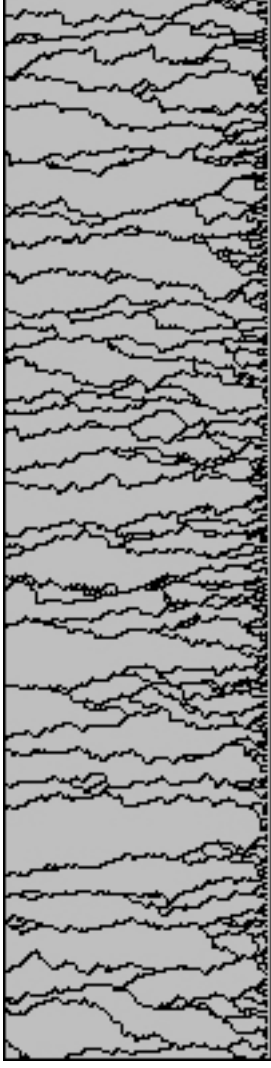
- High resistance of Zirconium to corrosion and hydrogen permeation is due to a specific “growth mode” texture
- One has to learn how to grow strong oxide texture and how to control the orientation of zirconium alloy substrate.



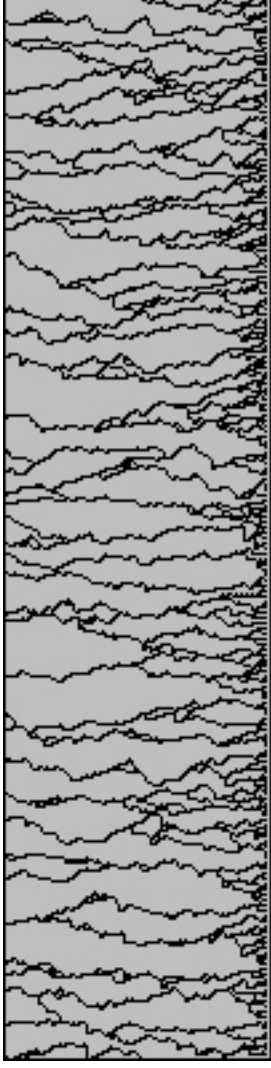
Application in Ni - NiO system

Oxide microstructure

On $\{100\}_{\text{Ni}}$



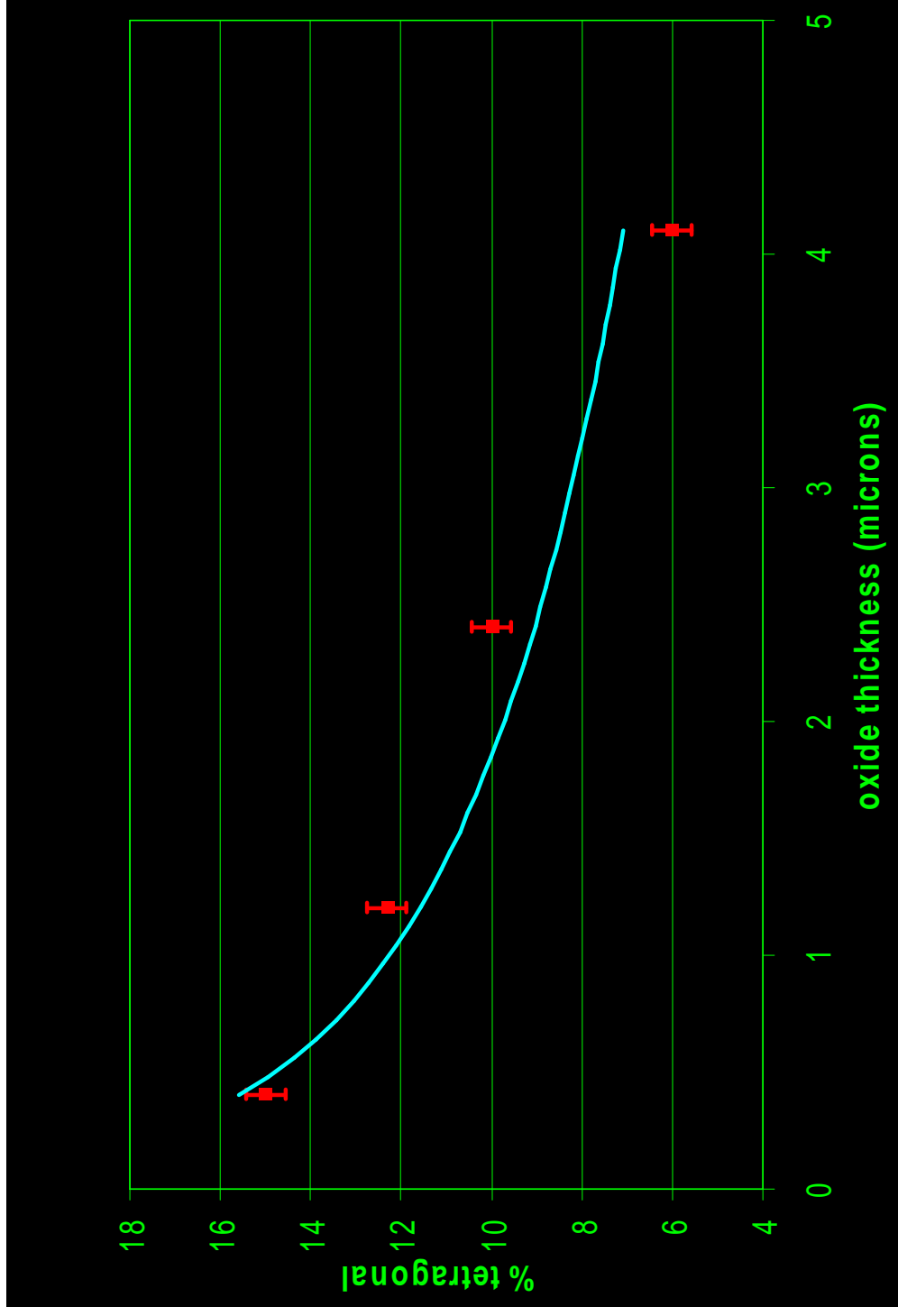
On $\{111\}_{\text{Ni}}$



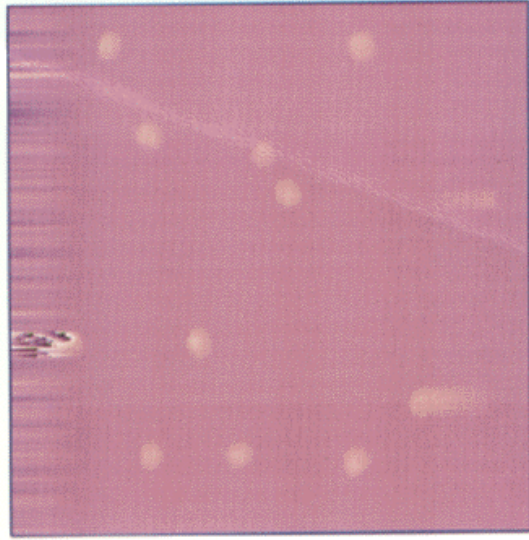
Experimental
microstructure



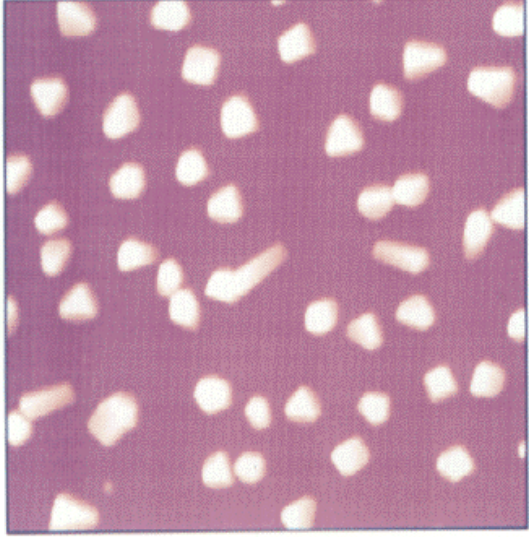
STEAM FORMED OXIDES



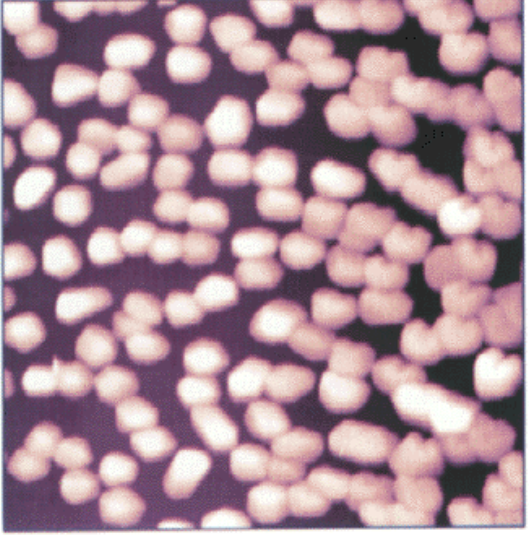
tmquen48.021



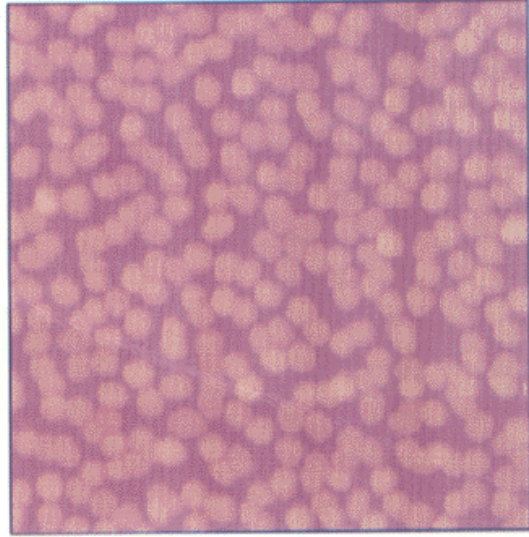
tmquen44.001



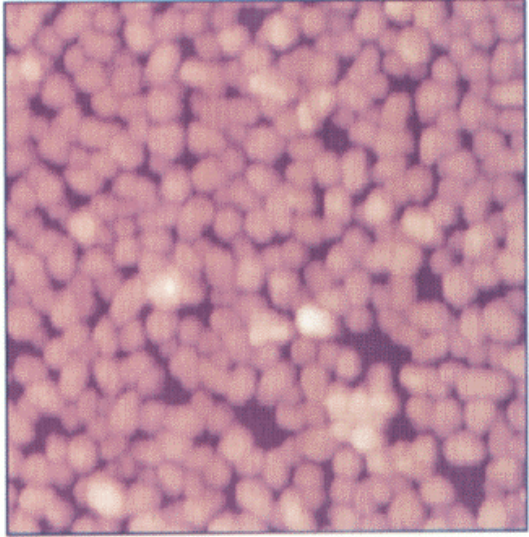
tmquen49.001



tmquen46.001



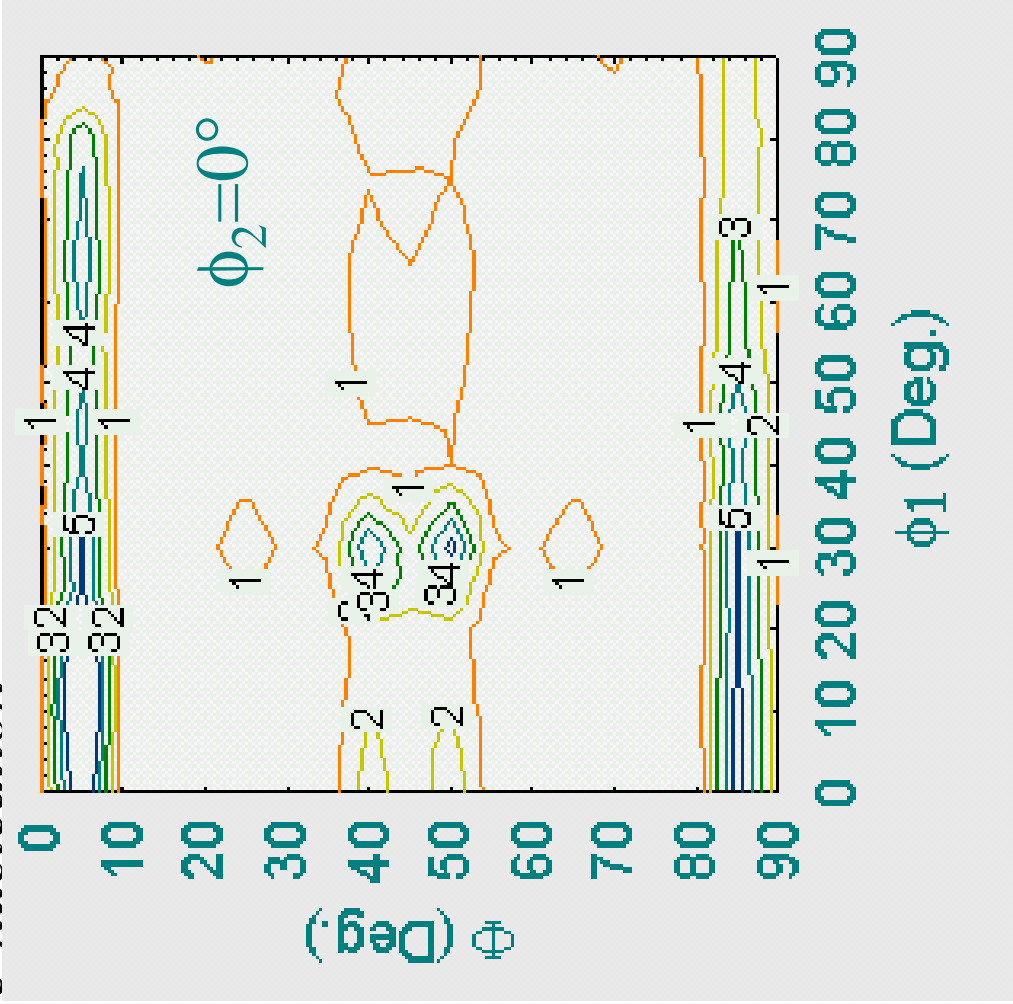
tmquen51.002



Application in Ni - NiO system

In the stage of growth and re-nucleation

| (hkl) | Surface energies (J/m^2) |
|---------|---------------------------------|
| (100) | 12.6 |
| (111) | 17.7 |
| (110) | 17.8 |
| (210) | 24.8 |
| (211) | 20.0 |
| (221) | 30.7 |



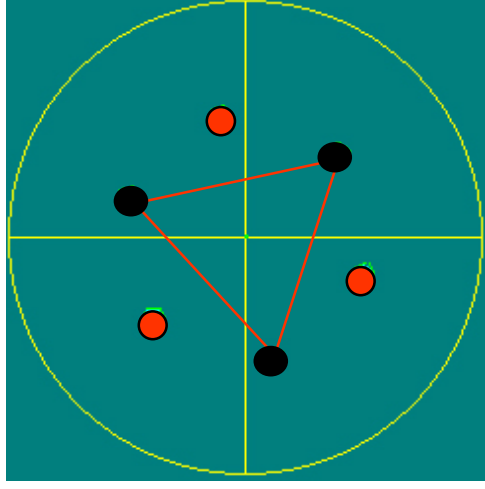
Calculated oxide ODF from experimental pole figures

Application in Ni - NiO system

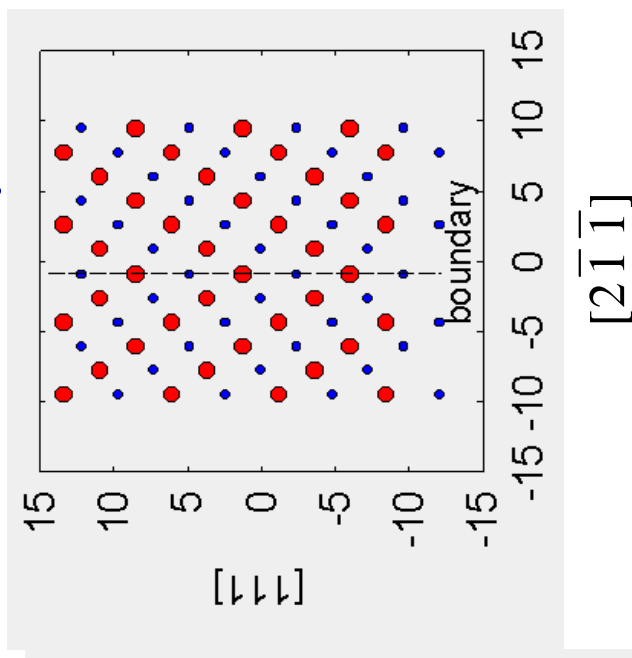
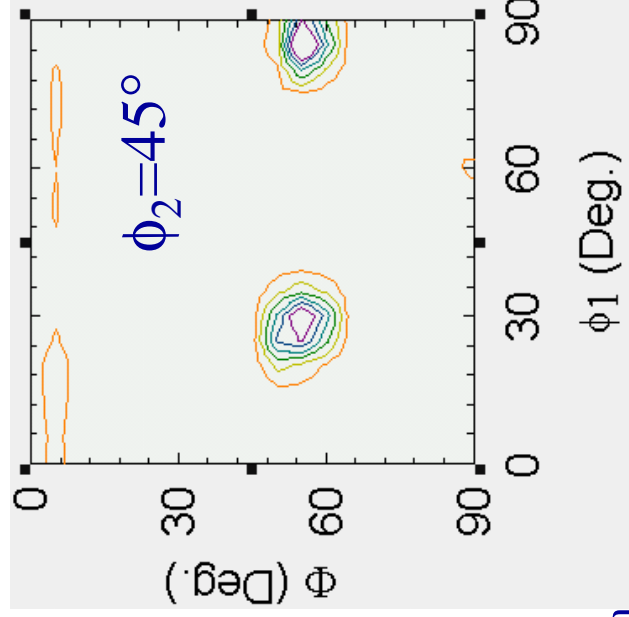
Effects of oxide texture and microstructure on the oxidation kinetics

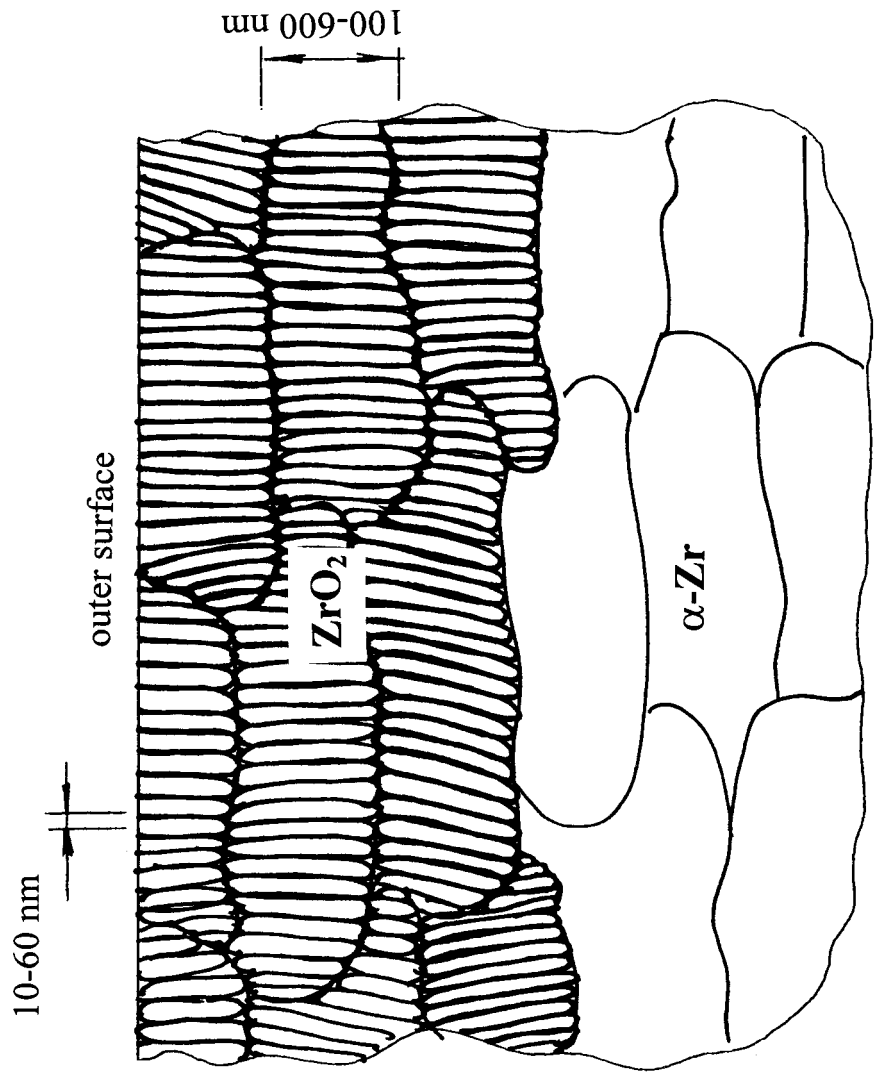
Grain boundary character distribution

- random grain boundary
- $\Sigma 3$ (twin) boundary (60° along $\langle 111 \rangle$)



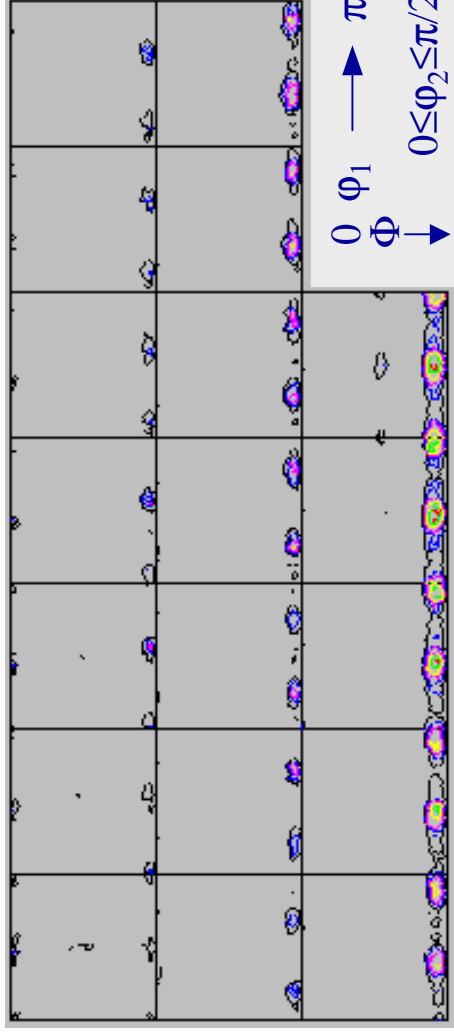
{100} pole figure of the oxide grown on {111}_{Ni}



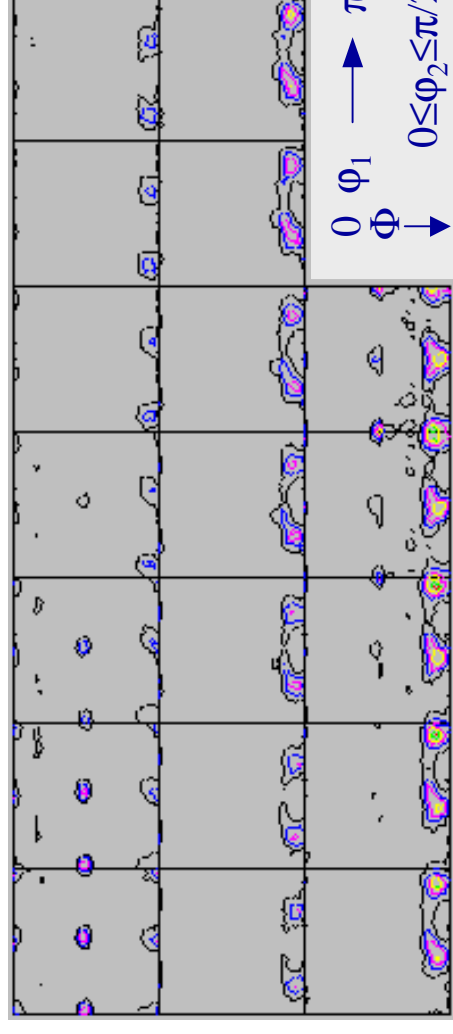


Application in Zr - ZrO₂ system

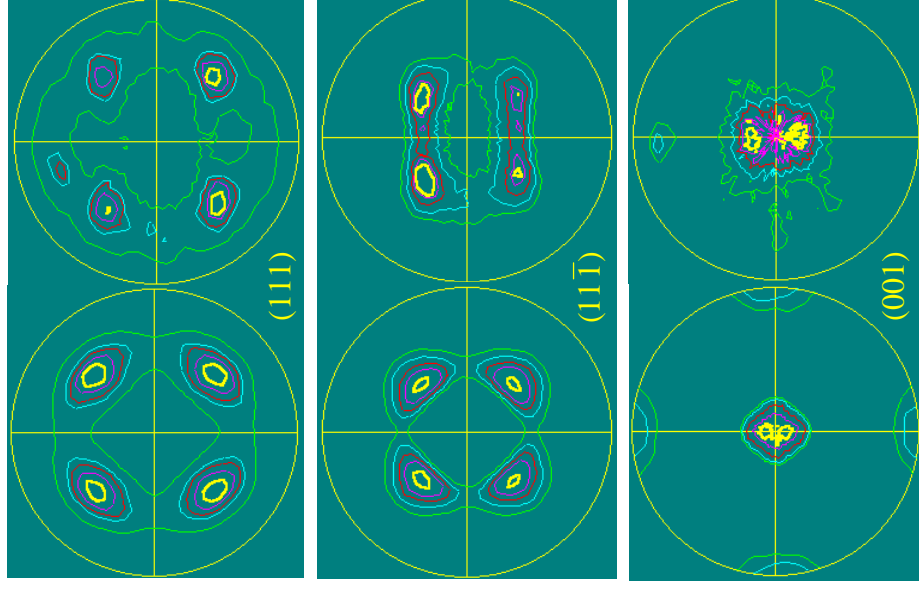
On RD sample



$0 \leq \varphi_1 \leq \pi$
 $0 \leq \varphi_2 \leq \pi/2$

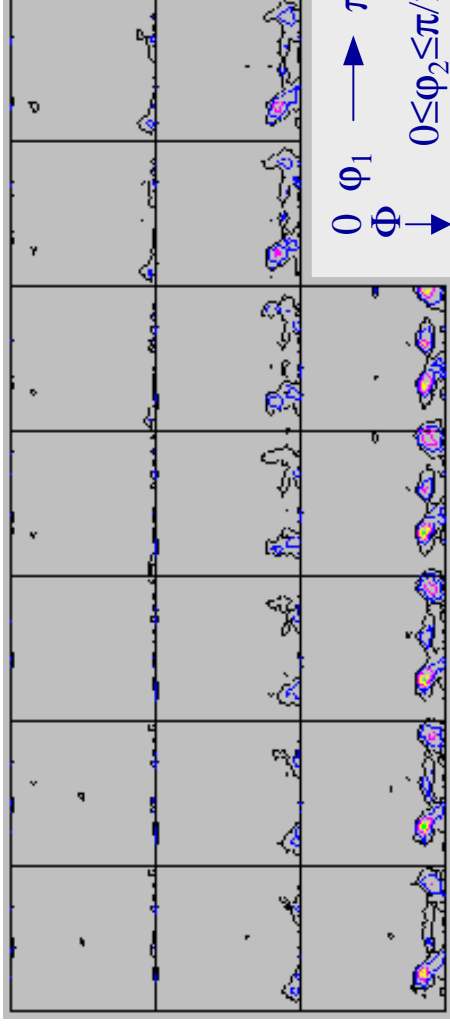
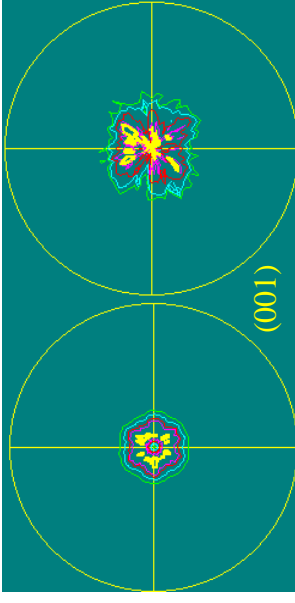
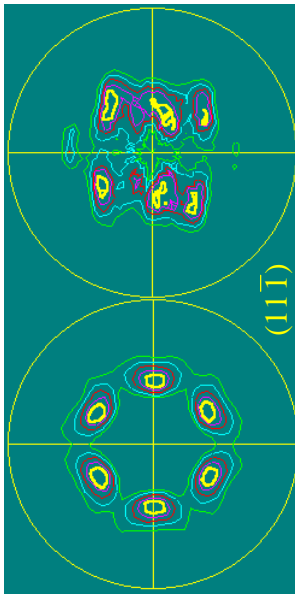
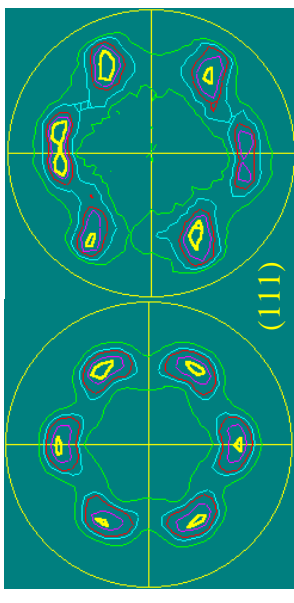
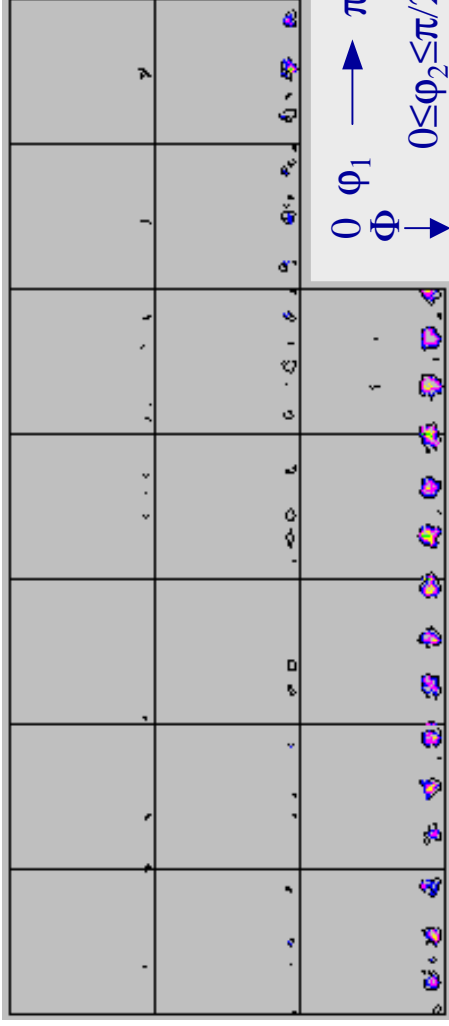


$0 \leq \varphi_1 \leq \pi$
 $0 \leq \varphi_2 \leq \pi/2$



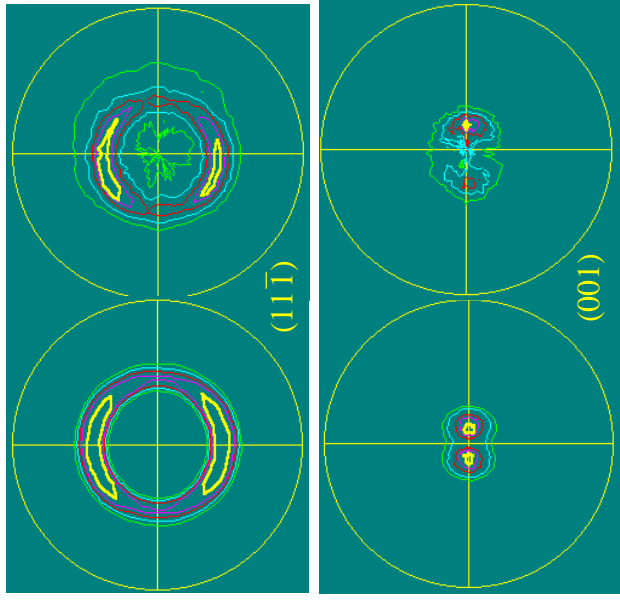
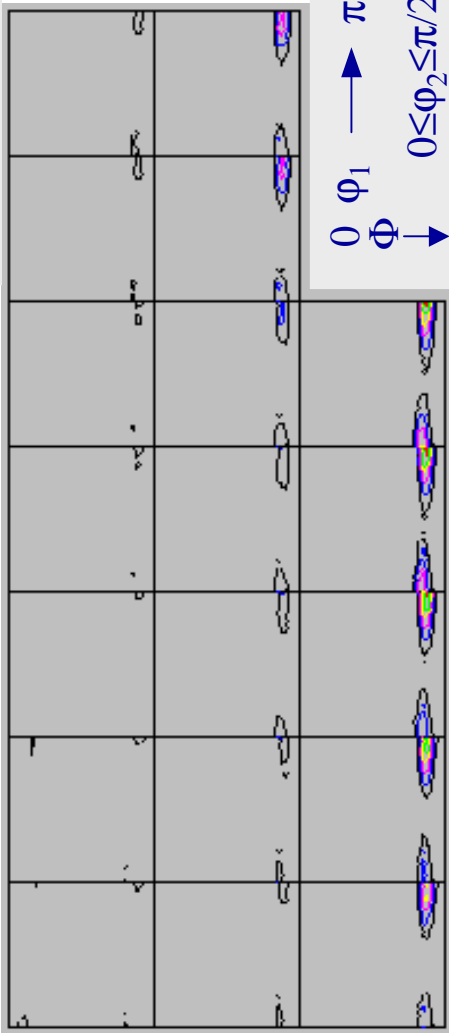
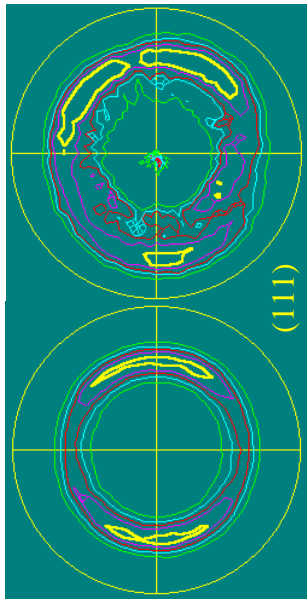
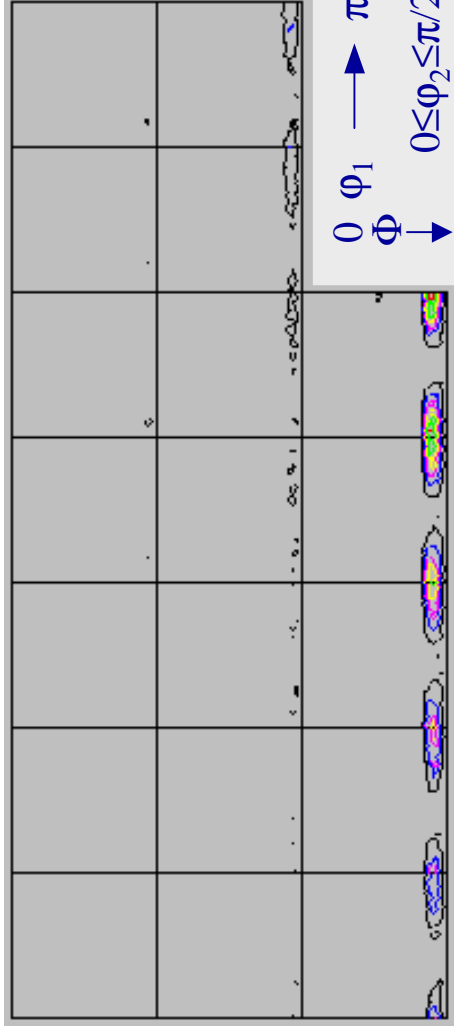
Application in Zr - ZrO₂ system

On TD sample



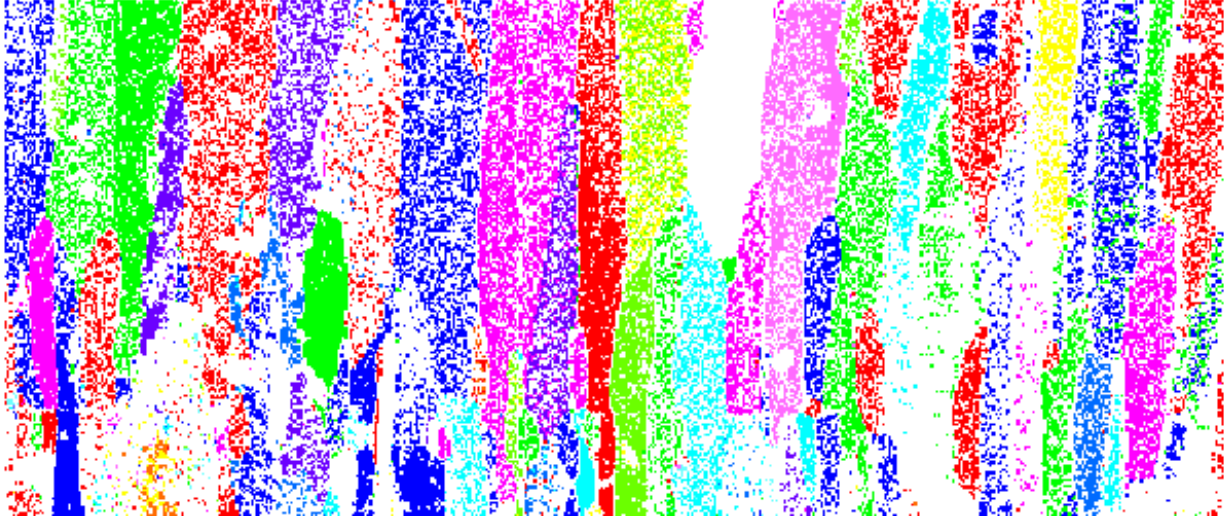
Application in Zr - ZrO₂ system

On LD sample



Contribution to the original knowledge

- 4) A mechanism of ZrO_2 texture and microstructure formation is proposed.
- 5) The diffusion mechanism in ZrO_2 is proposed.
- 6) The ODF of the oxide formed on the pressure tube is calculated for the first time.
- 7) A software is developed which can be used for the texture calculation in all the seven crystal systems



High Temperature Oxidation of Low Carbon Steel

Bae-Kyun Kim

Ph. D. Candidate



Dept. Mining and Metallurgical Eng. McGill
University

Supervisor: Prof. J.A. Szpunar

50 μm = 45 steps

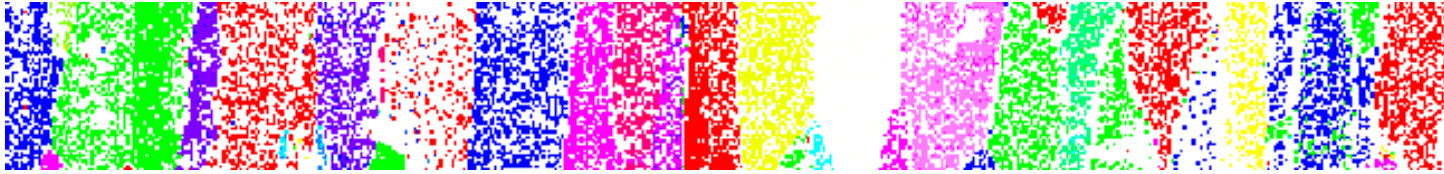
Oxidation Mechanism

The assumptions of the model of diffusion-controlled oxidation:

- The isothermal heating process above 570 ' .
- The isotropic oxide growth.

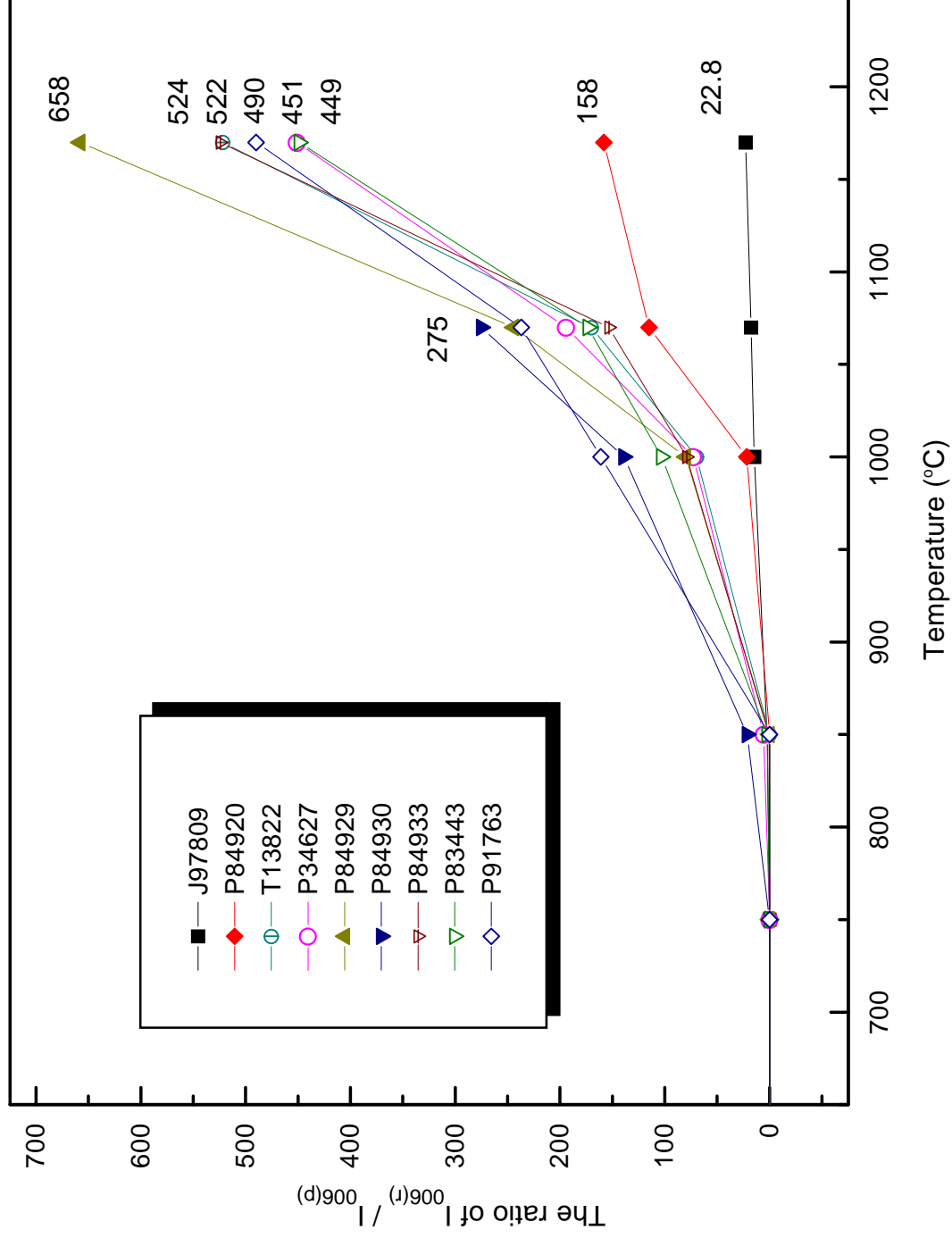
But, diffusion coefficient is one of anisotropic properties.

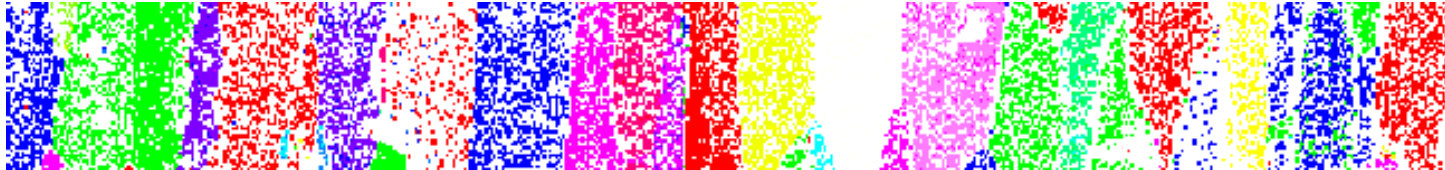
Industrial heat-treatment is a continuous heating.



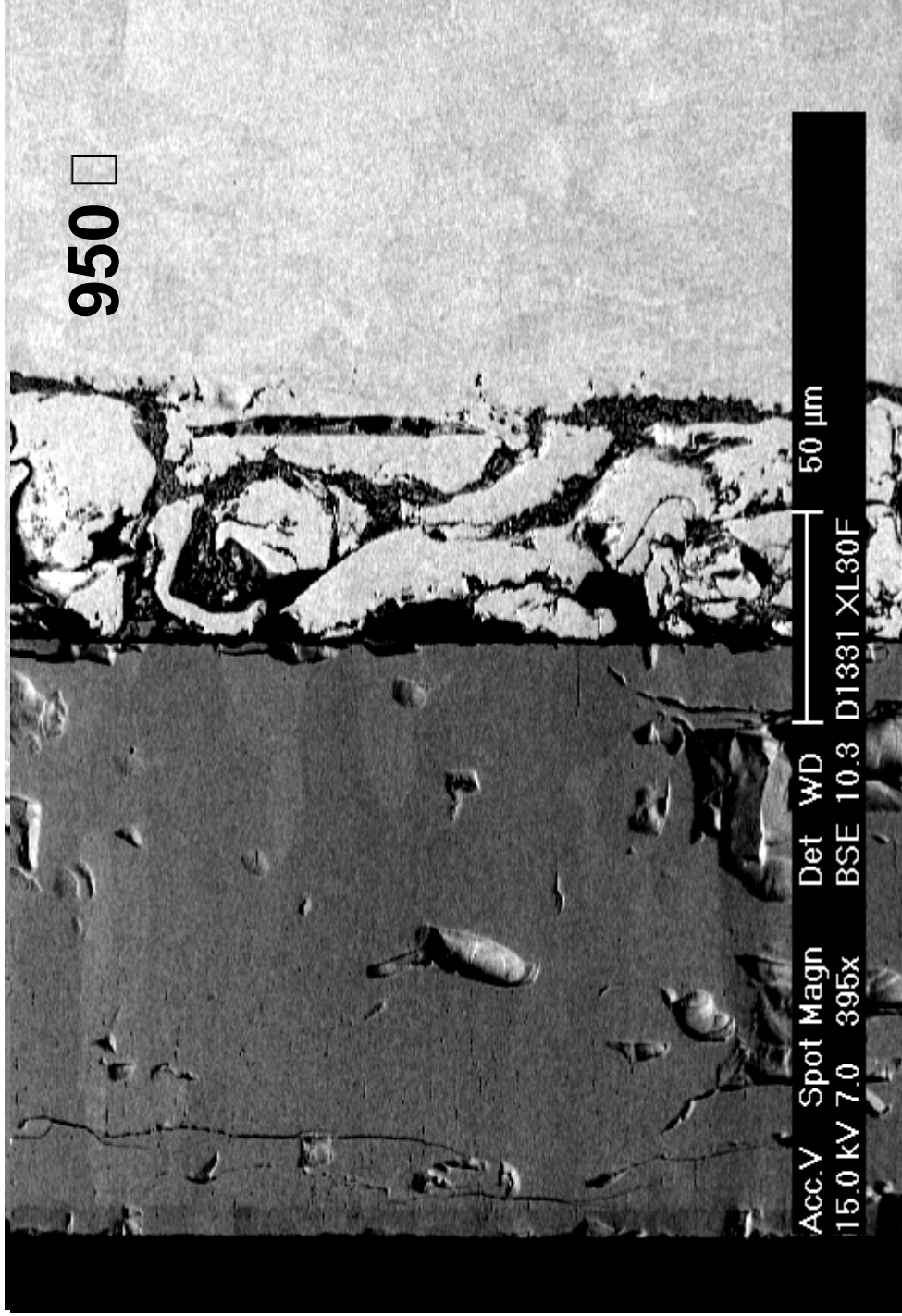
(006) of Hematite

The change of hematite (006) peak with temperature



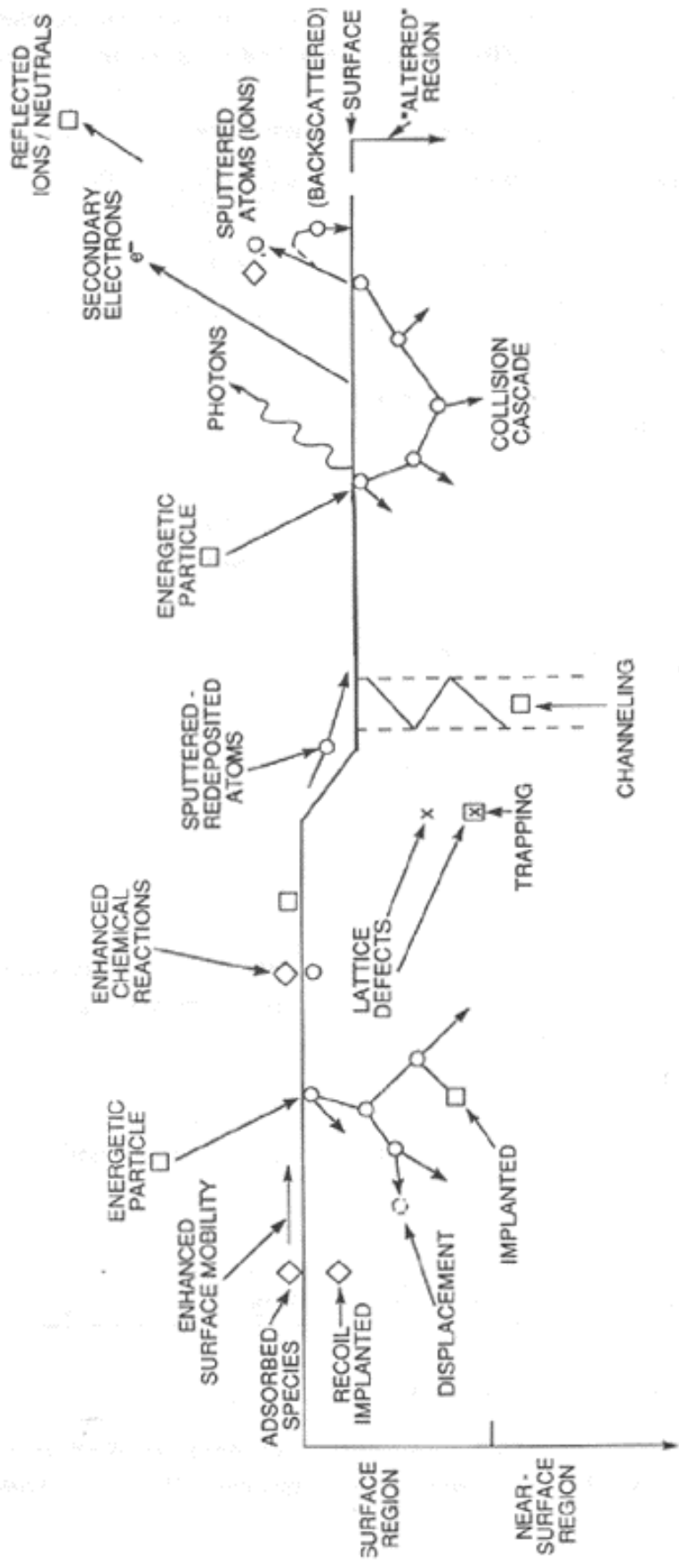


Microstructure of the oxides



BSE image after dynamic oxidation of P83443 in the tube furnace.

Sputtering: Depiction of Energetic Particle Bombardment Effects on Surface and Growing Films.



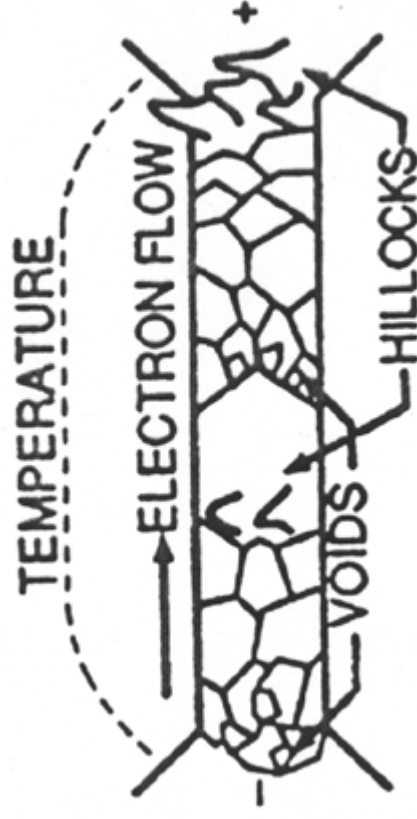
M. Ohring: The Materials Science of Thin Films

Texture and Electromigration Failure in Thin Films

* Electromigration

— migration of the metal atoms along the metallic conductor carrying large current densities (current density in interconnects $\geq 4 \times 10^5$ A/cm²)

* Damage (voids, hillocks) is formed by a depletion or accumulation of atoms.

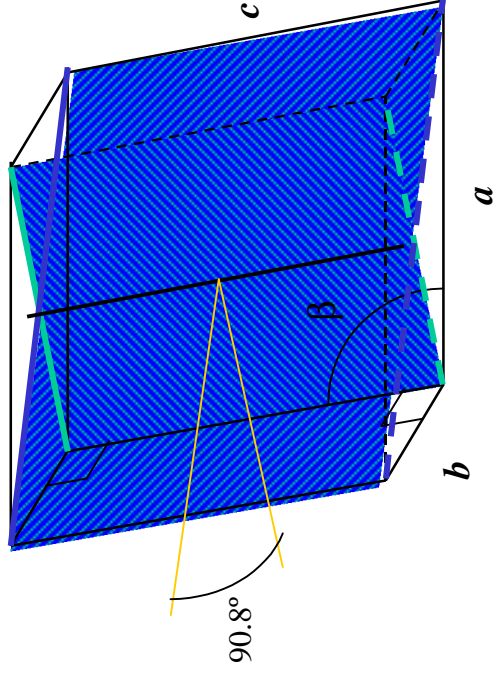


Table

Table 1
 Constants used in the calculation of the influence of the texture on the magnetic properties

| Legendre polynomials $\bar{P}_l(a)$ | Texture susceptibility coefficient λ_l | Texture remanence coefficient γ_l | Texture coercivity coefficient δ_{cl} | Texture anisotropy field δ_{Al} |
|-------------------------------------|--|--|--|--|
| $\bar{P}_0(a)$ | 0.943 | 0.707 | 0.530 | 1.672 |
| $\bar{P}_2(a)$ | -0.422 | 0.395 | 0.215 | -0.145 |
| $\bar{P}_4(a)$ | 0 | -0.088 | -0.082 | 0.223 |
| $\bar{P}_6(a)$ | 0 | 0.040 | 0.077 | -0.061 |
| $\bar{P}_8(a)$ | 0 | -0.023 | -0.011 | 0.032 |
| $\bar{P}_{10}(a)$ | 0 | 0.015 | 0.020 | -0.035 |
| $\bar{P}_{12}(a)$ | 0 | -0.010 | -0.005 | 0.051 |
| $\bar{P}_{14}(a)$ | 0 | 0.008 | 0.013 | -0.023 |
| $\bar{P}_{16}(a)$ | 0 | -0.006 | -0.002 | 0.033 |
| $\bar{P}_{18}(a)$ | 0 | 0.005 | 0.007 | -0.017 |
| $\bar{P}_{20}(a)$ | 0 | -0.004 | -0.002 | 0.024 |

Schematic of $\Sigma 71b$ misorientation in monoclinic ZrO_2



Formula for Texture Dependant Magnetic Properties

- Initial susceptibility

$$\chi = \frac{M^2}{2T} \sum_{l=0}^L t_l \lambda_l \quad ; T = \frac{1}{2} M^2 (N_T - N_0) + K$$

- Remanence

$$\frac{M_R}{M_S} = \sum_{l=0}^L t_l \bar{Y}_l$$

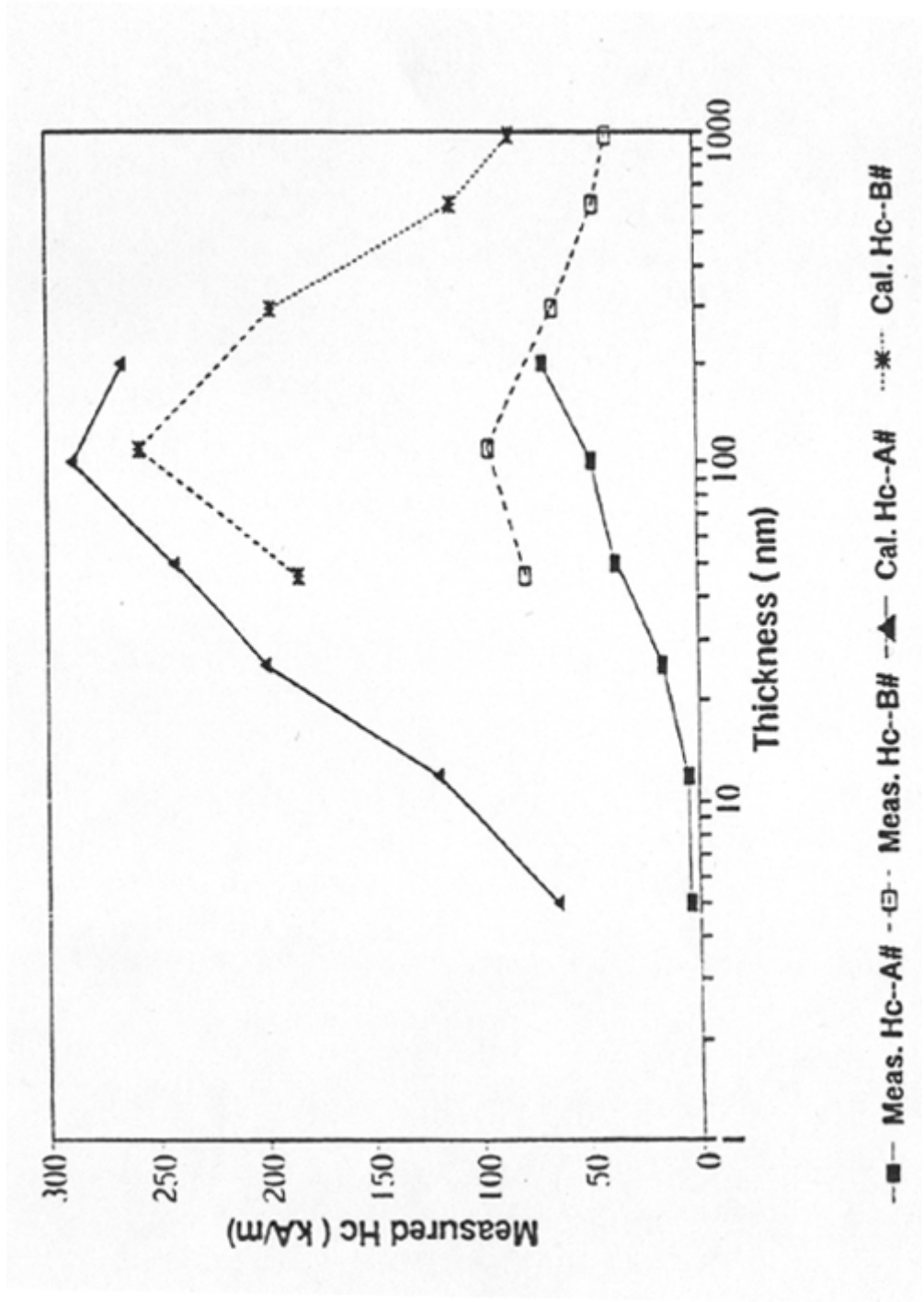
- The coercive force

$$H_C = \frac{2T}{M} \sum_{l=0}^L t_l \delta_{Cl}$$

- The anisotropy field

$$H_A = \frac{4T}{M} \sum_{l=0}^L t_l \delta_{Al}$$

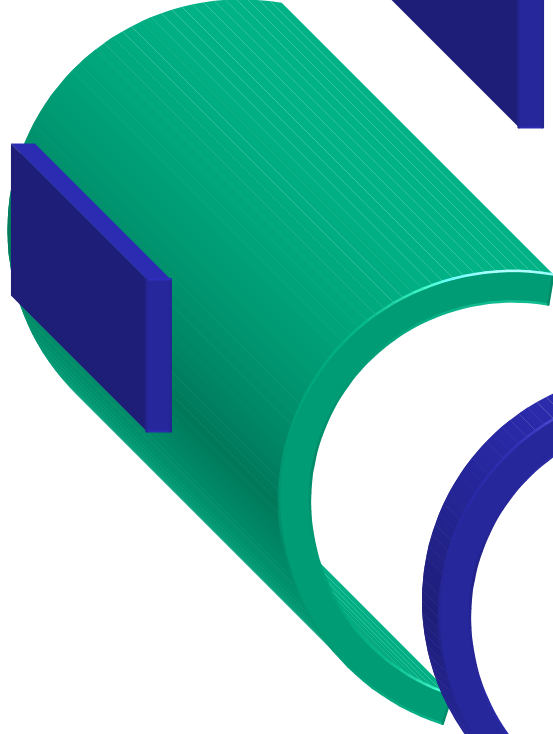
Comparison of the Measured and Calculated Coercivity



Application in Zr - ZrO₂ system

Selection of samples

Sample RD



Sample TD



Sample LD

